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# Surface Area-Enhanced Cerium and Sulfur-Modified Hierarchical Bismuth Oxide Nanosheets for Electrochemical Carbon Dioxide Reduction to Formate

Naveenkumar Palanimuthu,<sup>1</sup> Mohan Raj Subramaniam,<sup>1</sup> Muthu Austeria P.,<sup>2</sup> Preetam Kumar Sharma,<sup>3,4</sup> Vinoth Ramalingam,<sup>5</sup> Karthick Peramaiah,<sup>6</sup> Shanmugam Ramakrishnan,<sup>7</sup> Geun Ho Gu,<sup>2</sup> Eileen Hao Yu,<sup>4</sup> Dong Jin Yoo,<sup>1,8</sup>

- 1 Graduate School, Department of Energy Storage/Conversion Engineering (BK21 FOUR), Hydrogen and Fuel Cell Research Center, Jeonbuk National University, Jeonju, Jeollabuk-do, 54896, Republic of Korea
  - 2 Department of Energy Engineering, Korea Institute of Energy Technology (KENTECH), Naju, 58330, Republic of Korea
  - 3 Institute for Materials Discovery, University College London, Malet Place, London, WC1E 7JE, UK
  - 4 Department of Chemical Engineering, Loughborough University, Loughborough, LE11 3TU, UK
  - 5 School of Engineering, Robert Gordon University, Garthdee Road, Aberdeen, AB10 7GJ, UK
  - 6 Agency for Science, Technology, and Research, Institute of Sustainability for Chemicals, Energy and Environment, Singapore, 138634, Singapore
  - 7 School of Engineering, Newcastle University, Merz Court, Newcastle upon Tyne, NE17RU, UK
  - 8 Department of Life Science, Jeonbuk National University, Jeonju, Jeollabuk-do, 54896, Republic of Korea
- E-mail: djyoo@jbnu.ac.kr

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## Abstract

Electrochemical carbon dioxide reduction reaction (ECO<sub>2</sub>RR) is a promising approach to synthesize fuels and value-added chemical feedstocks while reducing atmospheric CO<sub>2</sub> levels. Here, high surface area cerium and sulfur-doped hierarchical bismuth oxide nanosheets (Ce@S-Bi<sub>2</sub>O<sub>3</sub>) are developed by a solvothermal method. The resulting Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst shows a maximum formate Faradaic efficiency of (FE) 92.5% and a current density of 42.09 mA cm<sup>-2</sup> at -1.16 V versus RHE using a traditional H-cell system. Furthermore, using a three-chamber gas diffusion electrode (GDE) reactor, a maximum formate FE of 85% is achieved in a wide range of applied potentials (-0.86 to -1.36 V vs RHE) using Ce@S-Bi<sub>2</sub>O<sub>3</sub>. The density functional theory (DFT) results show that doping of Ce and S in Bi<sub>2</sub>O<sub>3</sub> enhances formate production by weakening the OH\* and H\* species. Moreover, DFT calculations reveal that \*OCHO is a dominant pathway on Ce@S-Bi<sub>2</sub>O<sub>3</sub> that leads to efficient formate production. This study opens up new avenues for designing metal and element-doped electrocatalysts to improve the catalytic activity and selectivity for ECO<sub>2</sub>RR.

## Keywords

bismuth oxide; density functional theory; electrochemical CO<sub>2</sub> reduction; formate; surface area

## 1. Introduction

Increasing atmospheric carbon dioxide (CO<sub>2</sub>) levels from fossil fuel combustion and industrial activities is one of the most important global climate challenges.<sup>[1]</sup> Carbon capture and utilization technologies represent a potential solution for mitigating the CO<sub>2</sub> levels in the atmosphere.<sup>[2]</sup> Electrochemical CO<sub>2</sub> reduction (ECO<sub>2</sub>RR) is a promising strategy to transform CO<sub>2</sub> into a wide range of useful fuels and chemicals such as carbon monoxide, methane, formate, methanol, and ethanol.<sup>[3]</sup> Among various CO<sub>2</sub>-reductant products, formate is considered one of the most valuable products. Formate is an essential feedstock for the pharmaceutical industry, a hydrogen (H<sub>2</sub>) energy carrier for fuel cells, and fuel for direct electricity generation with high economic benefits.<sup>[4]</sup> Based on previously reported studies, several metal-based catalysts (e.g., Pd, In, Sn, Bi, Pb, Tl, Hg, and Cd), and metal oxides (e.g., In<sub>2</sub>O<sub>3</sub>, SnO<sub>2</sub>, Bi<sub>2</sub>O<sub>3</sub>, and CuO) have been reported to convert CO<sub>2</sub> to formate through ECO<sub>2</sub>RR.<sup>[5]</sup> However, many heavy metals (Pb, Cd, Hg, and Tl) are highly toxic and environmentally hazardous and are not recommended for practical applications.<sup>[6]</sup> In this regard, Bi-based electrocatalysts have been widely investigated due to their non-toxic nature, low-cost effectiveness, and earth abundance.<sup>[7]</sup>

Bi-based materials such as Bi<sub>2</sub>O<sub>3</sub><sup>[8]</sup> and Bi<sub>2</sub>S<sub>3</sub><sup>[9]</sup> have gained significant attention for their efficiency and selectivity in converting CO<sub>2</sub> to formate. However, most of these Bi electrocatalysts have limited activity (current density <40 mA cm<sup>-2</sup>) and low selectivity for formate (< 80%) due to the significance of H<sub>2</sub> evolution at higher current densities.<sup>[10]</sup> Numerous strategies have been introduced for Bi<sub>2</sub>O<sub>3</sub> catalyst design to improve the activity and selectivity of ECO<sub>2</sub>RR, such as nanostructures,<sup>[11]</sup> crystal facet control,<sup>[12]</sup> defect engineering,<sup>[13]</sup> tailoring the band structure,<sup>[14]</sup> metal doping,<sup>[15]</sup> and use with other materials. Based on that, modified Bi<sub>2</sub>O<sub>3</sub> hierarchical nanostructures have recently gained widespread acceptance due to their excellent catalytic performance.<sup>[16]</sup> Specifically, the nanostructure has a large specific surface area and abundant active sites that interact with the reactants and promote catalytic activity. The hierarchical nanostructure surfaces with high indices contain many coordinately unsaturated sites, edge sites, and structural defects, which play a crucial role in stabilizing key \*OCHO intermediate species.<sup>[17]</sup> This stabilization mechanism enhanced the catalytic activity of the Bi site, leading to increased production of formate.<sup>[18]</sup>

Moreover, modifying the electron density of the catalytic active site through heteroatom doping can further improve the catalytic efficiency of these nanostructures.<sup>[19–22]</sup> Specifically, the doped heteroatoms (such as N and S) can regulate the electron density of the catalyst, which modifies the electronic configurations of active sites to achieve the ideal adsorption energy with reaction intermediates for formate production in ECO<sub>2</sub>RR.<sup>[23]</sup> For instance, sulfur (S) doping creates a large number of oxygen vacancies and increases the adsorption intensity of CO<sub>2</sub>, which could enhance the electrocatalytic activity.<sup>[24]</sup> Similarly, metal atom doping in Bi<sub>2</sub>O<sub>3</sub> electrocatalysts has been explored for the ECO<sub>2</sub>RR to formate.<sup>[21,25,26]</sup> The inclusion of metal dopants can accelerate the charge transfer within the electrocatalyst, a crucial factor for optimizing the efficiency of electrochemical reactions and elevating reaction rates.<sup>[27]</sup> Metal doping has a role in modifying the electronic structure and surface defects, which is beneficial for producing formate in ECO<sub>2</sub>RR.<sup>[28]</sup> Recently, rare-earth-doped Bi<sub>2</sub>O<sub>3</sub> was used in ECO<sub>2</sub>RR. Yan et al., demonstrated that Eu-Bi doping resulted in enhanced selectivity and stability in a wide range of potentials in ECO<sub>2</sub>RR.<sup>[29]</sup> In a separate study, Zhou et al., demonstrated these benefits in the bimetal Ag-Bi-S-O catalyst, achieving exceptional formate selectivity >94% FE and a 12.5 mA cm<sup>-2</sup> current density with a wide range of potentials in ECO<sub>2</sub>RR.<sup>[30]</sup> Recently, the modified metal S-O has proved to be a better electrocatalyst for formate production in ECO<sub>2</sub>RR.<sup>[31]</sup> Furthermore, the selectivity in electrochemical CO<sub>2</sub> reduction depends on the relative confinement of the reaction intermediates (\*OCHO, \*HCOOH, \*CO, and \*H<sub>2</sub>) on the surface.<sup>[32]</sup> In this regard, modification strategies for optimizing the binding of intermediates on the electrocatalysts surface are essential for improving ECO<sub>2</sub>RR performance.<sup>[33]</sup>

In this work, we report the synthesis of Ce and S-modified hierarchical nanosheet Bi<sub>2</sub>O<sub>3</sub> electrocatalysts for the ECO<sub>2</sub>RR. The combined effect of Ce and S-doping, along with unique hierarchical nanosheet morphology, significantly enhances formate production. The prepared Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst displays Faradaic efficiency of formate (FE<sub>formate</sub>) > 90% in a wide potential window (-0.76 to -1.36 V vs RHE) in both analyzed H-cell and GDE reactor methods. Further, the density functional theory (DFT) calculations demonstrated the presence of Ce@S-Bi<sub>2</sub>O<sub>3</sub> and showed that the S and Ce modification reduced the energy barrier for forming the key \*OCHO intermediate, resulting in a highly optimized ECO<sub>2</sub>RR. This study offers a simple and effective strategy to design metal and heteroatom-doped Bi<sub>2</sub>O<sub>3</sub> electrocatalysts for ECO<sub>2</sub>RR.

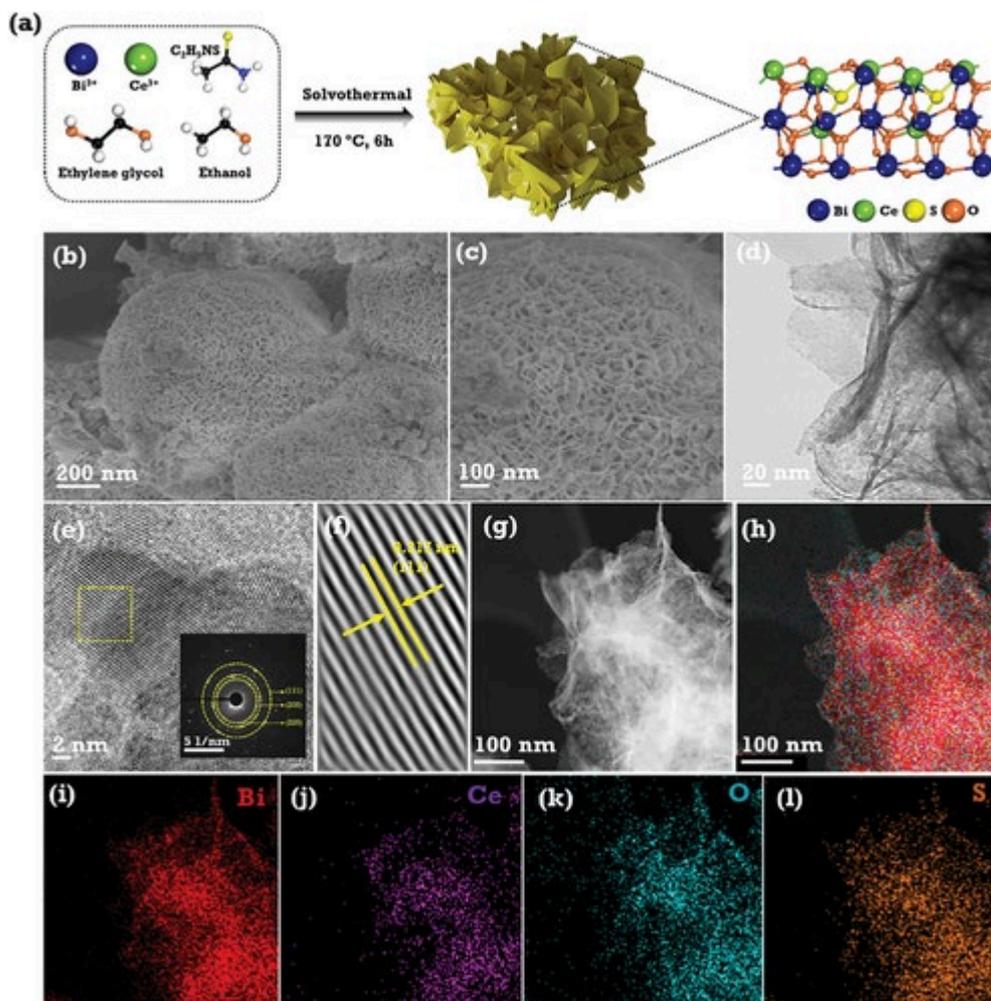
## 2. Results and Discussion

### 2.1 Structural Evaluation of the Synthesized Catalyst

Figure 1a illustrates a schematic representation of Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheet synthesis through a one-step solvothermal process. During the reaction process Bi<sup>3+</sup> ions in the metal precursor react with thioacetamide (S<sup>2-</sup>) ions and Ce<sup>3+</sup> ions to form hierarchical nanosheets of Ce@S-Bi<sub>2</sub>O<sub>3</sub>.<sup>[34]</sup> Briefly, bismuth (III) nitrate, cerium (III) nitrate, and thioacetamide were dissolved in a 1:2 ratio of ethylene glycol (EG) and ethanol solvent. The homogeneous solution was treated at 170 °C for 6 h by a solvothermal method. Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts were synthesized through a similar method. The prepared catalysts were used for ECO<sub>2</sub>RR.

**Figure 1**

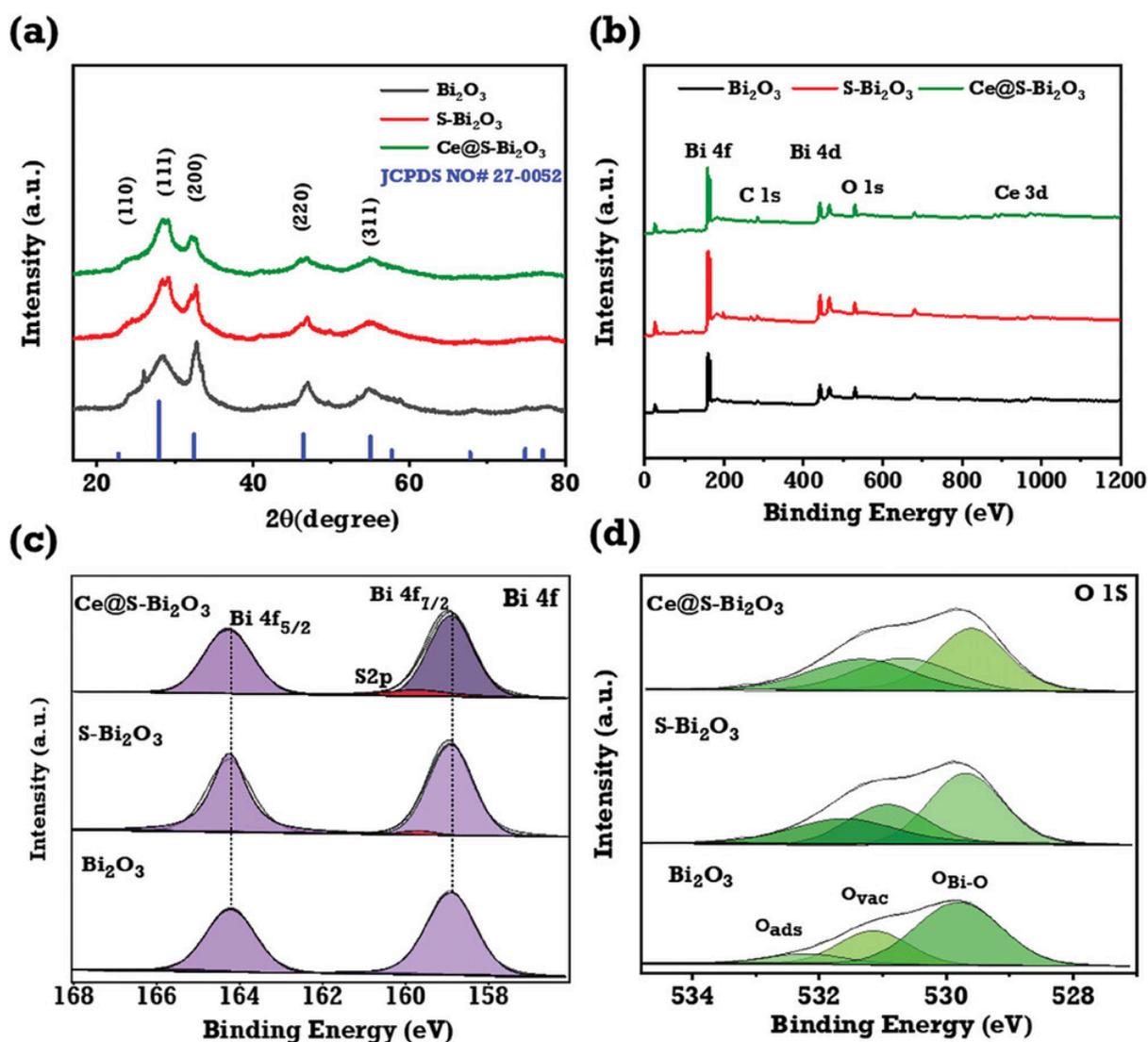
Synthesis and morphological characterization of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. Figure 1, a) Schematic illustration of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst synthesis process. b,c) FE-SEM images of Ce@S-Bi<sub>2</sub>O<sub>3</sub> demonstrate the hierarchical nanosheet morphology. d) TEM images of Ce@S-Bi<sub>2</sub>O<sub>3</sub> show that the hierarchical nanosheet. e,f) HR-TEM images and SAED pattern with crystal planes (111) of Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheet.



The morphology of the prepared catalysts was analyzed using FE-SEM technique. Figures S1 and S3 (Supporting Information) shows the FE-SEM images of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalysts and their corresponding energy dispersive X-ray spectroscopy (EDS) element maps. This reveals the formation of a 2–3 μm average size hierarchical microsphere structure in all synthesized catalysts with uniform distribution of the Bi, Ce, S, and O elements. Figure 1b,c shows an FE-SEM image of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst, which confirmed the formation of a hierarchical nanosheet structure upon addition of Ce and S with bare Bi<sub>2</sub>O<sub>3</sub>. Moreover, magnified FE-SEM images (Figure 1c) elucidated the origin of hierarchical microsphere growth, which occurred by assembling a hierarchical nanosheet structure with a uniform spacing of  $102.1 \pm 21.7$  nm. Further, the nanosheet assembly thickness was measured using ImageJ software and was in the range of  $10.41 \pm 3.05$  nm. During the solvothermal reaction, EG unambiguously directs the formation of a hierarchical nanosheet of Bi<sub>2</sub>O<sub>3</sub> due to its chelating ability.<sup>[35]</sup> When Bi(NO<sub>3</sub>)<sub>2</sub>·5H<sub>2</sub>O and Ce(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O are dissolved in EG, they chelate with glycols and form a respective alkoxide. Moreover, the

formation of these alkoxides can reduce the concentration of  $\text{Bi}^{3+}$  and  $\text{Ce}^{3+}$  ions in solution to form a uniform hierarchical nanosheet.<sup>[36]</sup> This phenomenon effectively prevents aggregation, facilitating the growth of even, well-defined hierarchical nanosheets with controlled thickness and desired properties. Furthermore, inductively coupled plasma-optical emission spectroscopy (ICP-OES) results reveal the doping percentages of Ce (3.3 at%) and S (1.2 at%) in the prepared  $\text{Ce@S-Bi}_2\text{O}_3$  catalysts, which agree with the EDS results from FE-SEM (Figure S3c, Supporting Information). We performed transmission electron microscopy (TEM) and high-angle annular dark field scanning TEM (HAADF-STEM) analysis to understand the hierarchical nanosheet formation in  $\text{Ce@S-Bi}_2\text{O}_3$ . The TEM image (Figure 1d) elucidates the presence of folded, nanosheet assemblies of  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst. A magnified TEM image (Figure S4b, Supporting Information) displays the formation of hierarchical nanosheets by orderly connected Ce and Bi nanoparticles, which induce the formation of a large nanosheet assembly. Some of the nanosheets were partially folded, and the fold lines exhibited darker features due to the accumulation of dense Bi and Ce metal nanoparticles, which led to the formation of hierarchical nanosheets.<sup>[37]</sup> An HR-TEM image (Figure 1e) shows the presence of highly crystalline catalysts with clear lattice fringes all over the catalysts. The associated FFT (fast Fourier transform) image shown in Figures 1f and S4 (Supporting Information) obtained from the HR-TEM image indicates that the obtained lattice fringes were related to the (111) plane with an interplanar spacing of 0.318 nm, which matches the formation of  $\text{Bi}_2\text{O}_3$  (JCPDS: 27-0052).<sup>[38]</sup> Notably, (HAADF-STEM) examination showed that these distinct nanosheet interactions and overlapping interactions were largely consistent, leading to the formation of foldable nanoparticles attached to the surface of hierarchical nanosheets represented in Figure 1g,h.<sup>[24]</sup> However, some discontinuities might exist in the inherited lattice defects, while oxide layers appeared due to partial oxidation of the Bi catalyst.<sup>[39]</sup> STEM-EDS mapping demonstrated the uniform distribution of Bi and Ce and S, O elements along the hierarchical nanosheet as represented in Figure 1i-l, and (Figure S5, Supporting Information). The hierarchical nanosheets provide a platform for efficient  $\text{CO}_2$  conversion by providing a larger number of active site and enhanced mass transport so it is more beneficial for catalytic activity surfaces in  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalysts. Moreover, the hierarchical nanosheet structure enhances catalytic performance by exposing a substantial surface area with numerous active sites and enabling easy electrolyte penetration and diffusion in electrochemical activity.<sup>[40]</sup> Also, HAADF-STEM shows a hierarchical nanosheet folded edge in the image (Figure S6, Supporting Information). These  $\text{Bi}_2\text{O}_3$  nanosheets exposed large quantities of nanoparticles arranged in nanosheets.<sup>[37,41]</sup> This is accompanied by breaking of Bi-O bonds and the formation of Bi nanosheets. As a result, structural defects in bismuth can significantly improve its electrochemical performance.<sup>[39,42]</sup> Moreover, Brunauer-Emmett-Teller (BET) surface area measurements of the synthesized  $\text{Bi}_2\text{O}_3$ ,  $\text{S-Bi}_2\text{O}_3$ , and  $\text{Ce@S-Bi}_2\text{O}_3$  (obtained using a Microtrac, BELsorp-mini II) from nitrogen-adsorption-desorption isotherms are shown in Figure S7a-c (Supporting Information). The hierarchical nanosheets of sulfur-modified  $\text{S-Bi}_2\text{O}_3$  and  $\text{Ce@S-Bi}_2\text{O}_3$  exhibited high surface areas of ( $44.01 \text{ m}^2 \text{ g}^{-1}$ ) and ( $50.48 \text{ m}^2 \text{ g}^{-1}$ ) compared to undoped  $\text{Bi}_2\text{O}_3$  ( $30.40 \text{ m}^2 \text{ g}^{-1}$ ). The abruptly improved surface area (>40%) would be expected to improve catalyst reaction kinetics during the  $\text{ECO}_2\text{RR}$ . The high-surface-area mesoporous S-doped  $\text{Bi}_2\text{O}_3$  might be due to the presence of  $\text{SO}_x$  gases evolved during the solvothermal reactions with addition sulfur source ( $\text{C}_2\text{H}_5\text{NS}$ ).<sup>[43]</sup> Further, the obtained mesoporous pore diameter (21.52 nm) and enhanced pore volume ( $0.2716 \text{ cm}^3 \text{ g}^{-1}$ ) (inset Figure S7a-c and

Table S1, Supporting Information) clearly revealed the formation of highly active Ce@S-Bi<sub>2</sub>O<sub>3</sub> porous materials in this reaction. All the active sites in Bi are available to the reactant due to the high mesoporosity, which could enhance charge transfer rate during the electrochemical reaction.<sup>[34]</sup>



**Figure 2**

Crystal structure and chemical states of the attained (Ce@S-Bi<sub>2</sub>O<sub>3</sub>) electrocatalyst. a) XRD patterns of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> of electrocatalysts. b) High-resolution survey spectrum of (Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst). c) High-resolution XPS spectra of Bi 4f. d) O 1s XPS spectrum of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst.

The crystal patterns of the synthesized Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets were determined by XRD analysis (Figure 2a). Diffraction peaks noted at 22.74°, 27.94°, 32.38°, 46.44°, and 55.08° correspond to (1 1 0), (1 1 1), (2 0 0), (2 2 0), and (3 1 1) lattice planes, respectively, which confirmed that the prepared Bi<sub>2</sub>O<sub>3</sub> had a cubic phase with a space group Pn3m.<sup>[28]</sup> The predominant peak  $\approx 27.94^\circ$  corresponds to the (111) plane of Bi<sub>2</sub>O<sub>3</sub>. The observed XRD results agree with the JCPDS number of 27-0052, indicating the formation of the cubic phase  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>.<sup>[38]</sup> Moreover, after undergoing S-doping, the peaks became broader. These specific

changes can be attributed to the reduced crystallinity. This indicates that the S modification led to a less ordered and reduced crystallite size in the S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. Apart from  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> phase, no additional peaks appeared in S-Bi<sub>2</sub>O<sub>3</sub>, suggesting the existence of either a small proportion or amorphous nature of S within the S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst.<sup>[43b]</sup> Furthermore, the electrochemical study indicates the enhanced catalytic activity of the amorphous nature of S-doped Bi<sub>2</sub>O<sub>3</sub> compared to the undoped electrocatalyst. Moreover, when S was added to Bi<sub>2</sub>O<sub>3</sub>, the peaks in the crystal structure were separated, possibly due to S having a similar lattice parameter to  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>. This analysis further confirmed the importance of S-doping in the Bi<sub>2</sub>O<sub>3</sub> nanostructures. The slight positive shift of  $2\theta$  ensures the incorporation of Ce in S-Bi<sub>2</sub>O<sub>3</sub>.<sup>[44]</sup> No diffraction reflections other than those for  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> were detected, which indicated the formation of a highly pure  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> phase.

Furthermore, X-ray photoelectron spectroscopy (XPS) measurements were carried out to investigate the elemental compositions and chemical states of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst as shown in Figure 2b. Figure 2c presents the Bi 4f<sub>7/2</sub> and Bi 4f<sub>5/2</sub> peak pair at 159.0 and 164.2 eV for Bi<sub>2</sub>O<sub>3</sub>, which was ascribed to Bi species. Compared to as-synthesized Bi<sub>2</sub>O<sub>3</sub>, a positive shift of approximately 0.3 eV was observed for the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst.<sup>[45]</sup> The XPS peak corresponding to S 2p peak 159.8 eV shifts slightly toward higher binding energies, indicating enhanced S bond strength. This observation is highlighted the ability of heteroatom doping to effectively modulate the bonding strength of materials.<sup>[46]</sup> Additionally, the S-doped Bi<sub>2</sub>O<sub>3</sub> spectrum has a peak between core Bi at 159.7 eV. For S-Bi<sub>2</sub>O<sub>3</sub> a slight peak of Bi 4f<sub>7/2</sub> was observed and was assigned to S<sup>2-</sup> in sulfides.<sup>[47]</sup> This suggests that S was present at the weaker O sites of Bi<sub>2</sub>O<sub>3</sub>, resulting in S bonding with the Bi site (Bi-S). Furthermore, the incorporation of S was shifted to 0.3 eV. Moreover, the Ce<sup>3+</sup> ions and oxygen vacancy sites were located on the surface of Ce. The Ce@S-Bi<sub>2</sub>O<sub>3</sub> showed a positive shift of 0.18 eV in the Bi sites compared with Bi<sub>2</sub>O<sub>3</sub>, indicating partial charge from Bi to Ce, and the electronic structure of Bi was tuned by the strong interaction with Ce.<sup>[45]</sup> Figure S8 (Supporting Information) shows the deconvoluted Ce 3d core level spectrum, which revealed the valence states of both the Ce<sup>3+</sup> and Ce<sup>4+</sup> multiple d-splitting such as 3d<sub>3/2</sub> and 3d<sub>5/2</sub>. The characteristic peaks of the mixed valence, such as Ce<sup>4+</sup> 3d<sub>5/2</sub> and Ce<sup>4+</sup> 3d<sub>3/2</sub>, were located at binding energies of 916.7 and 898.6 eV, respectively. The Ce<sup>3+</sup> ions consist of peaks located at binding energies of 901.6 and 881.3 eV, which were assigned to Ce<sup>3+</sup> 3d<sub>3/2</sub> and Ce<sup>3+</sup> 3d<sub>5/2</sub>, respectively.<sup>[48]</sup> The mixed states of Ce ions were also indicated by additional satellite peaks (orbital “shake-up”) located at 06.8 eV for Ce<sup>3+</sup> 3d<sub>3/2</sub> and 885.3 and 889.2 eV for Ce<sup>3+</sup> 3d<sub>5/2</sub>.<sup>[49,50]</sup> The Ce 3d spectrum indicates that the concentration of Ce<sup>3+</sup> ions increased, and their lattice parameters expanded because of the reduced electrostatic force derived from the increased concentration of Ce<sup>3+</sup> ions. This result indicates the presence of Ce<sup>3+</sup>, which could include O-vacancies.<sup>[51–53]</sup> A decrease in lattice oxygen intensity was observed in the S and Ce-doped Bi<sub>2</sub>O<sub>3</sub> electrocatalyst.

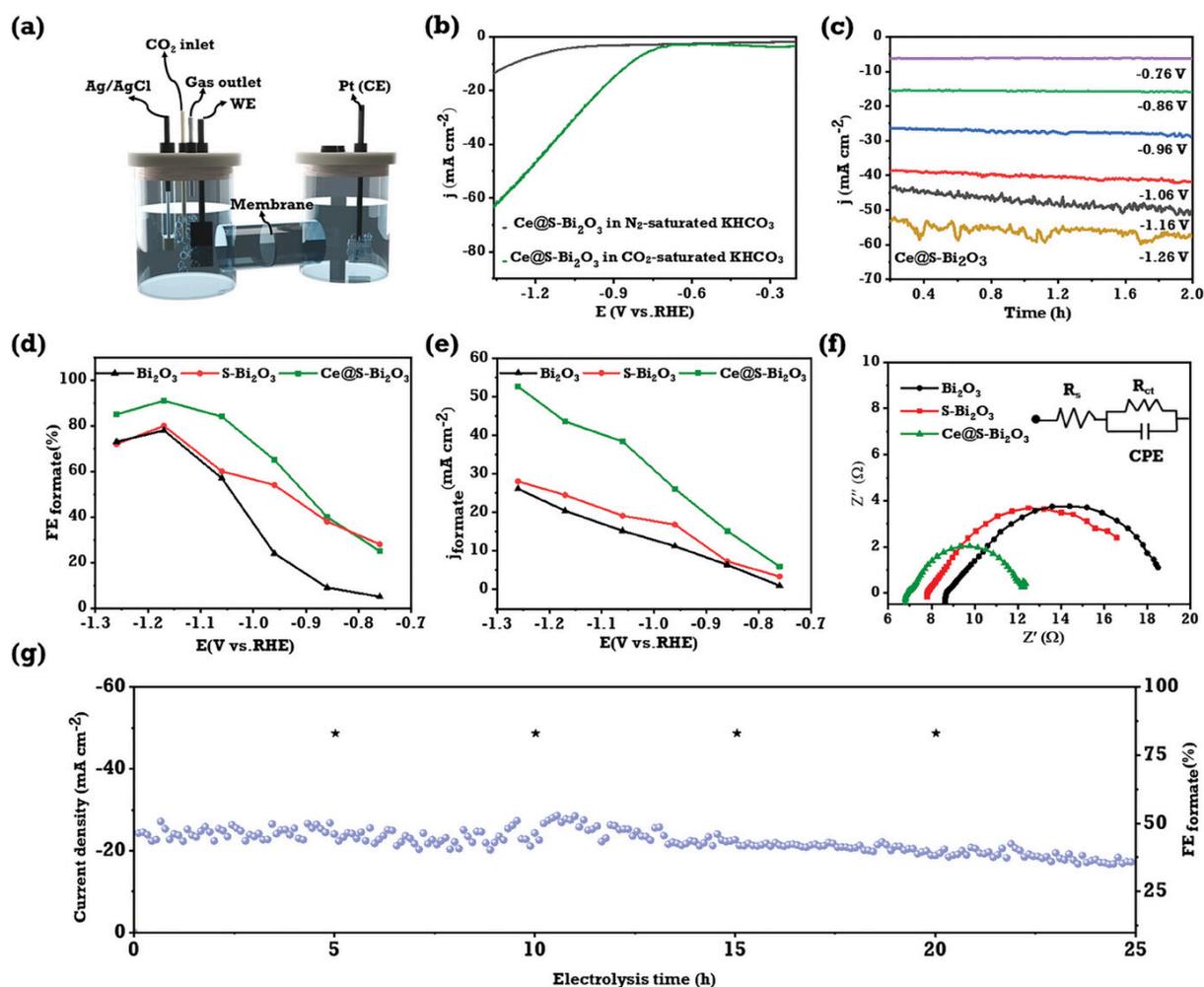
The O 1s spectrum of the Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub> and Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst can be deconvoluted into three peaks denoted as Bi-O bond (O<sub>lat</sub>), an oxygen vacancy (O<sub>vac</sub>), and adsorbed oxygen (O<sub>ads</sub>) characteristics of oxygen atoms bonded to metals peaks at 529.8, 529.7, and 529.6 eV, respectively. Interestingly, the characteristic peaks of Ce@S-Bi<sub>2</sub>O<sub>3</sub> shown at 531.1, 530.9, and 530.7 eV (O<sub>vac</sub>) decreased compared to those of S-Bi<sub>2</sub>O<sub>3</sub> at 530.7 eV. The high concentrations of O-vacancies and Ce<sup>3+</sup> ions may enhance the electrocatalytic selectivity for facilitated adsorption of CO<sub>2</sub> with O-bonding.<sup>[51a]</sup> The adsorbed oxygen (O<sub>ads</sub>) peaks are located at 532.1, 531.6, and 531.3 eV in Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst, respectively, and the oxygen vacancy peak of 530.7 eV can be ascribed to the presence of Ce@S-Bi<sub>2</sub>O<sub>3</sub> as

shown in Figure 2d. Furthermore, Fourier transform infrared (FTIR) and Raman spectroscopy analysis were performed to understand the surface properties of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst. The FTIR spectra of Ce@S-Bi<sub>2</sub>O<sub>3</sub> shows the peaks at 400 to 600 cm<sup>-1</sup> and 1047, 1355, 2159, and 2972 cm<sup>-1</sup> are the characteristic groups of Bi—O, respectively (Figure S9a, Supporting Information).<sup>[54]</sup> The peak at 3380 cm<sup>-1</sup> belongs to the adsorbed water and surface hydroxyl peaks O—H, extending vibrations over the surface of the electrocatalyst.<sup>[55]</sup> Furthermore, the peak edge at 879 cm<sup>-1</sup> is assigned to the Bi—O—C bending vibrations of Bi<sub>2</sub>O<sub>3</sub> in the presence of air adsorbed.<sup>[56]</sup> In addition, the Raman characteristic bands at ≈980 to 1200 cm<sup>-1</sup> are ascribed to the doping of S presence in Bi—O. The band broadening at ≈731 and 1046, 1248, and 1571 cm<sup>-1</sup> belongs to Ce-O.<sup>[45,57]</sup> Hence, it is confirmed that incorporating S and Ce changes the surface area of the Bi<sub>2</sub>O<sub>3</sub> nanosheet. To further understand the structure of the catalyst, we performed Raman analysis Figure S9b (Supporting Information). The peaks noted at 149192 and 307 cm<sup>-1</sup> corresponds to the Bi—O outline representing various bending and stretching modes in the Bi<sub>2</sub>O<sub>3</sub> nanosheet.<sup>[58]</sup> From the spectrum, the slight peak present in 465 cm<sup>-1</sup> represents the Ce—O and, further the broad, penetrating peaks ≈437 cm<sup>-1</sup> are typical peaks of Ce—O.<sup>[59]</sup> In addition, the peak observed at 254 and 970 cm<sup>-1</sup> is attributed to the S in Bi<sub>2</sub>O<sub>3</sub> nanosheet.<sup>[60]</sup>

## 2.2 Electrochemical CO<sub>2</sub> Reduction Reactions

Electrochemical CO<sub>2</sub>RR was carried out using an H-cell, as illustrated in Figure 3a. Further the Figures 3b and S10 (Supporting Information) show the linear sweep voltammetry (LSV) curves of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts in N<sub>2</sub> and CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolyte with a scan rate of 5 mV s<sup>-1</sup>. The current density obtained for all three electrocatalysts in the CO<sub>2</sub>-saturated electrolyte was significantly higher than that in N<sub>2</sub>-saturated conditions, which shows efficient ECO<sub>2</sub>RR activity.<sup>[61]</sup> Moreover, the obtained LSV curves of S-Bi<sub>2</sub>O<sub>3</sub> and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts show a higher current density and positive onset potential than Bi<sub>2</sub>O<sub>3</sub>. Significantly, the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst was outperformed with more onset potential (-1.16 V vs RHE) and higher current density in wide potential ranges compared to other investigated electrocatalysts.<sup>[17, 62, 63]</sup> Notably, for instance, Ce@S-Bi<sub>2</sub>O<sub>3</sub> showed a current density of 42.07 and 6.01 mA cm<sup>-2</sup> at -1.16 V versus RHE in CO<sub>2</sub> and N<sub>2</sub>-saturated electrolyte, respectively, a difference of 6.8 times higher compared to N<sub>2</sub>-saturated conditions (Figure 3b). To clarify the performance of ECO<sub>2</sub>RR, chronoamperometry (CA) curves (i-t) were obtained across various potentials ranging from -0.76 to -1.26 V versus RHE as shown in Figure 3c. Furthermore, the ECO<sub>2</sub>RR intrinsic activity of Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts was analyzed by a CA technique for 2 h at different applied potentials. The produced liquid product formate was identified using proton nuclear magnetic resonance spectroscopy (<sup>1</sup>H-NMR), as shown in Figure S11 (Supporting Information). Figure 3d shows the FE of formate for Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts. In that, the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst achieved FE<sub>formate</sub> over 92.5% at -1.16 V versus RHE, and it has a much higher formate selectivity compared to the Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub>, which had FE<sub>formate</sub> values of 80% and 83%, respectively, at the same potentials. Among all three examined electrocatalysts, the Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets exhibited higher current density (49.9 mA cm<sup>-2</sup>) compared to Bi<sub>2</sub>O<sub>3</sub> (25.3 mA cm<sup>-2</sup>) and S-Bi<sub>2</sub>O<sub>3</sub> (27.02 mA cm<sup>-2</sup>) at -1.16 V versus RHE, as shown in Figure 3e. Also, the achieved higher FE<sub>formate</sub> and j<sub>formate</sub> of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst is comparable to the recently reported Bi-based electrocatalysts in an H-cell (Table S6, Supporting Information). In addition, electrocatalysts display much higher formate partial current densities

( $j_{\text{formate}}$ ) in the different potentials during CA analysis, as shown in Figure S12a,b (Supporting Information).



**Figure 3**

ECO<sub>2</sub>RR performance of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. a) Schematic diagram of H-cell. b) LSV curve of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst under N<sub>2</sub> and CO<sub>2</sub> saturated 0.5 M KHCO<sub>3</sub> electrolyte with a scan rate of 5 mV s<sup>-1</sup>. c) Chronoamperometry measurements of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst at different applied potentials (from -0.76 to -1.26 V vs RHE) for electrochemical CO<sub>2</sub>RR. d) FE<sub>formate</sub> of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. e) The partial current density of formate ( $j_{\text{formate}}$ ) for Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts. f) Electrochemical impedance Spectra of ECO<sub>2</sub>RR at the corresponding electrocatalyst, insert are equivalent circuits used for fitting, and g) Stability tests of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts at -0.96 V versus RHE in 0.5 M KHCO<sub>3</sub> electrolyte.

In addition, the Tafel slope ( $\log(j \text{ mA cm}^{-2})$  vs overpotential) as shown in Figure S13a–c (Supporting Information) the Ce@S-Bi<sub>2</sub>O<sub>3</sub> (257 mV dec<sup>-1</sup>) has a lower Tafel slope compared to Bi<sub>2</sub>O<sub>3</sub> (286 mV dec<sup>-1</sup>) and S-Bi<sub>2</sub>O<sub>3</sub> (273 mV dec<sup>-1</sup>), which confirmed the improved reaction kinetics of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst over the other un-doped electrocatalysts in ECO<sub>2</sub>RR.<sup>[64]</sup> Also, the electrochemical active surface area was estimated from the double-layer capacitance ( $C_{dl}$ ) to understand the activity of the prepared electrocatalyst during ECO<sub>2</sub>RR. The results in

Figure S14a–d (Supporting Information) show the  $C_{dl}$  values of  $\text{Bi}_2\text{O}_3$ ,  $\text{S-Bi}_2\text{O}_3$ , and  $\text{Ce@S-Bi}_2\text{O}_3$  (0.94, 0.87, and 1.12  $\text{mF cm}^{-2}$ , respectively). ECSA was calculated from the  $C_{dl}$  values, and the results show that  $\text{Ce@S-Bi}_2\text{O}_3$  exhibits a relatively higher active surface area of 28.0  $\text{cm}^{-2}$  compared to  $\text{Bi}_2\text{O}_3$  (23.5  $\text{cm}^{-2}$ ) and  $\text{S-Bi}_2\text{O}_3$  (21.7  $\text{cm}^{-2}$ ). This indicates that the highly efficient active surface area sites and mass transfer of  $\text{Ce@S-Bi}_2\text{O}_3$  are more suitable for  $\text{ECO}_2\text{RR}$ .<sup>[65]</sup> Furthermore, the turn over frequency (TOF) was calculated for  $\text{Bi}_2\text{O}_3$  (0.0234  $\text{S}^{-1}$ ),  $\text{S-Bi}_2\text{O}_3$  (0.0296  $\text{S}^{-1}$ ), and  $\text{Ce@S-Bi}_2\text{O}_3$  (0.05624  $\text{S}^{-1}$ ) at the corresponding current potential (−1.16 V vs RHE). These results further indicate that the  $\text{Ce@S-Bi}_2\text{O}_3$  hierarchical nanosheets are favorable for creating more abundant active sites on the electrocatalyst surface for improving the  $\text{ECO}_2\text{RR}$  activity. Electrochemical impedance spectroscopy (EIS) was carried out to investigate the electron kinetics charge transfer resistance as shown in Figure 3f. The charge transfer resistance ( $R_{ct}$ ) of  $\text{Ce@S-Bi}_2\text{O}_3$  (3.3  $\Omega$ ) shows better conductivity than that of  $\text{S-Bi}_2\text{O}_3$  (3.6  $\Omega$ ) and  $\text{Bi}_2\text{O}_3$  (4.2  $\Omega$ ) electrocatalysts, as demonstrated in the Nyquist plots. The corresponding fitted data using the equivalent circuit diagram are shown in insert Figure 3f. The lower  $R_{ct}$  values indicate better electronic conductivity and charge transfer process of the  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst. These specific findings showed that the hierarchical nanosheets facilitate faster charge carrier migration to the  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst surface.<sup>[66–68]</sup> To evaluate the stability of the hierarchical nanosheet  $\text{Ce@S-Bi}_2\text{O}_3$  electrode, CA analysis was performed at −0.96 V versus RHE for >25 h. The current density was stable at  $\approx 21 \text{ mA cm}^{-2}$ , and the FE of formate remained unchanged ( $86\% \pm 2\%$ ) during CA analysis, as demonstrated in Figure 3g. The S has lower electronegativity than oxygen (O), thus reducing the Bi oxidation state in  $\text{S-Bi}_2\text{O}_3$  electrocatalyst. The lower amount of S-doping enhanced the catalytic activity by modifying the surface electronic structure Bi-metal sites. Also, the S-doping decreased the energy barrier of intermediate and obstructed the adsorption  $^*\text{H}$ , which suppressed the evaluation of  $\text{H}_2$ .<sup>[69,46]</sup> Moreover, doping of S with Ce in  $\text{Bi}_2\text{O}_3$  further enhanced the formate selectivity due to its better intrinsic activity. Specifically, the Ce doping created more numerous active Bi sites and facilitated the dissociation of  $\text{H}_2\text{O}$  to form adsorbed  $\text{H}^*$  intermediate, eventually enhancing  $\text{FE}_{\text{formate}}$  during  $\text{ECO}_2\text{RR}$ .<sup>[31,45]</sup> In addition, role of Ce and S-doping in  $\text{Bi}_2\text{O}_3$  was theoretically investigated and discussed in DFT section.

### 2.3 Post Electrochemical Analysis

Additionally, FE-SEM and HR-TEM analyses were carried out after 2 h of the  $\text{ECO}_2\text{RR}$  to understand the structural and electrochemical stability of the  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst. The results in Figures S15 and S16 (Supporting Information) show the appreciable surface structure and intensity of the elements in EDX analysis. HR-TEM images in Figure S15a,b (Supporting Information) show the retention of catalyst elements distributed uniformly without agglomeration. After  $\text{ECO}_2\text{RR}$  the electrocatalyst exhibits the d-spacing value of Bi (0.323 nm) corresponds to the (012) plane well match with metallic  $\text{Bi}^0$ , which indicate the conversion of  $\text{Bi}^{3+}$  to metallic  $\text{Bi}^0$ .<sup>[10]</sup> Also, the SAED pattern in Figure S16c (Supporting Information) illustrates the improved crystallinity nature of  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst. The EDS mapping in Figure S16d,e (Supporting Information) confirmed the uniform distribution of elements in the hierarchical nanosheets. Respective elemental maps confirmed the presence of Ce, S, and O elements with decreases in their respective concentration. Figure S16f (Supporting Information) shows the corresponding EDS spectrum with an appreciable intensity of all the elements and the metal composition in the selected mapping area. To understand the structural state of catalysts during the  $\text{ECO}_2\text{RR}$  process, we have performed time-dependent XRD analysis and shown in Figure S17a (Supporting Information). The analysis is performed under different reaction time at an applied potential of −1.16 V versus RHE. For the  $\text{Ce@S-Bi}_2\text{O}_3$  electrode, with the increasing reaction time of  $\text{ECO}_2\text{RR}$  (from 120 s), the diffraction peaks of  $\text{Bi}_2\text{O}_3$  (JCPDS#27-

0052) reduce gradually. Furthermore, the diffraction peaks of Bi (JCPDS#85-1330) and  $\text{Bi}_2\text{O}_2\text{CO}_3$  (JCPDS#41-1488) appear completely after 2 h.<sup>[70,71]</sup> Furthermore XPS analysis performed after  $\text{ECO}_2\text{RR}$  and shown in Figure S17b (Supporting Information) to understand the elements state and chemical composition of the catalyst. In that, the Bi  $4f_{7/2}$  and Bi  $4f_{5/2}$  peaks initially at 159.2 and 164.5 eV were shifted to 158.1 and 163.3 eV (Bi  $4f_{7/2} = 1.11$  eV and Bi  $4f_{5/2} = 1.12$  eV), which agree with the obtained d-spacing values from the HR-TEM results. The reduced  $\text{Bi}^{3+}$  electrons are more beneficial for  $\text{CO}_2$  to formate conversion than their initial nature.<sup>[72,73,8]</sup> After  $\text{ECO}_2\text{RR}$  Figure S17c (Supporting Information) revealed that the peak located at 530.2 eV in the O1s spectrum belongs to the Bi–O ( $\text{O}_{\text{lat}}$ ), and the adsorbed oxygen peak ( $\text{O}_{\text{ads}}$ ) is located at 532.8 eV. After electrochemical  $\text{CO}_2$  reduction, the O-vacancy sites were entirely reduced to metallic Bi species.<sup>[74,10]</sup> Furthermore, Figure S17d and Table S2 (Supporting Information) revealed that, in Ce 3d, the valence state of  $\text{Ce}^{4+}$  ions gradually reduced to  $\text{Ce}^{3+}$  ions, which are more favorable for formate selectivity during the  $\text{ECO}_2\text{RR}$ . As shown in Table S2 (Supporting Information), we have performed ICP-OES analysis for  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst before and after  $\text{ECO}_2\text{RR}$  to study the changes in the composition of Bi, Ce, and S. The electrocatalyst after  $\text{ECO}_2\text{RR}$  showed a nominal decrease in the concentrations of Bi, Ce and S compared to fresh electrocatalyst. This decrease in the concentrations may be due to the leaching during a prolonged stability test.<sup>[46]</sup> However, this does not affect the current density and selectivity of electrocatalyst throughout the stability test as evidenced from Figure 3g. Finally, the post  $\text{ECO}_2\text{RR}$  investigations revealed that the optimized  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst displays structural retention and formation of a promising chemical nature.

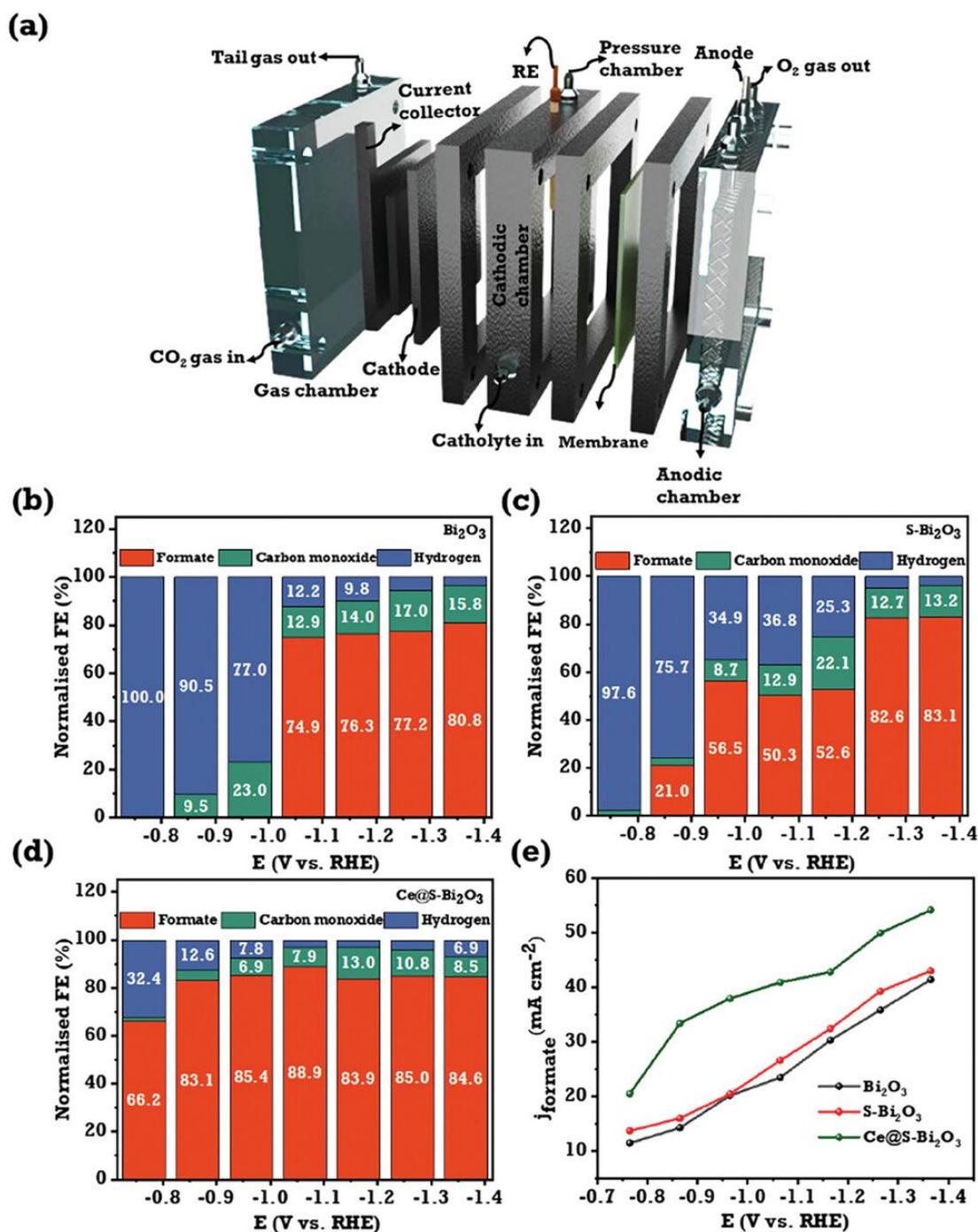
## 2.4 Electrochemical $\text{CO}_2$ Reduction in GDE Reactor

Evaluating  $\text{ECO}_2\text{RR}$   $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalysts in three-chamber GDE reactors is an effective way to screen electrocatalysts, understand structure-performance relationships, and gain mechanistic insights, as in Figure 4a.<sup>[75]</sup> The  $\text{Ce@S-Bi}_2\text{O}_3$  represents a more positive onset potential and increases the current density compared to  $\text{Bi}_2\text{O}_3$  and  $\text{S-Bi}_2\text{O}_3$ , suggesting it is a suitable catalyst for  $\text{ECO}_2\text{RR}$ . The obtained electrochemical reduction products corresponding to different applied potentials ( $-0.76$  to  $-1.36$  eV vs RHE) were measured using online gas chromatography and ion chromatography. The much higher  $\text{FE}_{\text{formate}}$  was obtained for  $\text{Ce@S-Bi}_2\text{O}_3$  than for  $\text{S-Bi}_2\text{O}_3$  and  $\text{Bi}_2\text{O}_3$  electrocatalyst in the analyzed potential region from  $-0.76$  to  $-1.36$  eV versus RHE. Notably, the highest  $\text{FE}_{\text{formate}}$  (88.9%) was obtained at  $-1.06$  eV versus RHE ( $40.90 \text{ mA cm}^{-2}$ ) in the  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst. As shown in Figure 4d, formate is the dominant product from the electrochemical  $\text{CO}_2$  reduction using  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst.  $\text{CO}$  and  $\text{H}_2$  gases were also detected as minority products in  $\text{ECO}_2\text{RR}$ . The  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst demonstrated high FE ( $> 80\%$ ) toward formate in a wide potential window (of  $-0.76$  to  $-1.36$  eV) versus RHE. At a lower cathodic potential of  $-0.76$  eV, significant  $\text{H}_2$  evolution takes place with a corresponding FE of 32.4%. Upon obtaining the potential, the  $\text{H}_2$  evolution is suppressed, and  $\text{CO}$  contribution starts to increase and reaches a maximum  $\text{FE}_{\text{CO}}$  of 13% at  $-1.16$  eV. Notably, the S containing S-doped  $\text{Bi}_2\text{O}_3$  and  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalysts enhanced the dissociation of  $\text{H}_2\text{O}$  and increase the  $\text{H}^+$ , which might react with  $\text{CO}_2$  to form  $\text{OCHO}^*$ , suppresses  $\text{H}_2$  formation and promotes formate selectivity at low overpotentials ( $-0.76$  to  $-1.36$  eV vs RHE) in  $\text{ECO}_2\text{RR}$ .<sup>[76]</sup> Also compared to single metal Bi-S-O catalysts Ce doped Ce-Bi-S-O catalyst exhibit more superior catalytic performance in the formate selectivity, it is clearly confirm from the obtained results shown in Figure 4d. At a higher potential of  $-1.36$  eV,  $\text{H}_2$  starts to rise again. The reason for enhanced  $\text{H}_2$  evolution and at

a higher potential is the mass transport limitation of CO<sub>2</sub> to the electrode at higher currents. The current densities also increased with an increase in the potential. Although the FE at -1.36 V was slightly lower than at -1.26 V (due to the higher current at -1.36 V), the partial current density of formate is higher (54.13 mA cm<sup>-2</sup>) than the 49.93 mA cm<sup>-2</sup> for -1.26 V versus RHE under a CO<sub>2</sub> atmosphere increase in order, this results from the enhanced conductivity of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst shown in Figure 4d. In addition, gas phase products were measured through an online gas chromatogram, and no gas products other than CO and H<sub>2</sub> were observed on gas chromatography. The presence of formate is identified from the ion chromatogram from the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. The products for S-Bi<sub>2</sub>O<sub>3</sub> and Bi<sub>2</sub>O<sub>3</sub> catalysts were also determined at -1.36 eV versus RHE. The FE<sub>formate</sub> were slightly lower, with values of 83 and 81%, respectively. The partial current density toward formate for Bi<sub>2</sub>O<sub>3</sub> was 41.4 mA cm<sup>-2</sup> and that for S-Bi<sub>2</sub>O<sub>3</sub> was 43.0 mA cm<sup>-2</sup>, which is much lower than that toward Ce@S-Bi<sub>2</sub>O<sub>3</sub>, as shown in Figure 4b,c. Through these experiments, it was determined that Ce@S-Bi<sub>2</sub>O<sub>3</sub> is the best catalyst in the sequence for selective formate production by ECO<sub>2</sub>RR. Notably, Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst delivers a higher partial current density for formate than Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub>. Ce doped at the S-Bi<sub>2</sub>O<sub>3</sub> provides a high formate current density in ECO<sub>2</sub>RR. Moreover, all the investigated catalysts (Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub>) showed the highest j<sub>formate</sub> achieved at -1.06 eV versus RHE. Furthermore, the stability of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst was evaluated in a GDE flow reactor at an applied potential of -1.2 V versus RHE, for a continuous reaction of up to 5 h. Figure S18b (Supporting Information) shows that there is no significant change in the current density during the stability analysis. The FE of formate is also sustained at over 72% for up to 4 h of reaction shown in Figure S18a (Supporting Information). However, after 4 h, the FE of formate decreased to 65% and H<sub>2</sub> FE increased to 28%, which could be due to the flooding of GDE.<sup>[77]</sup> A similar phenomenon was observed in our previous study as well.<sup>[78]</sup>

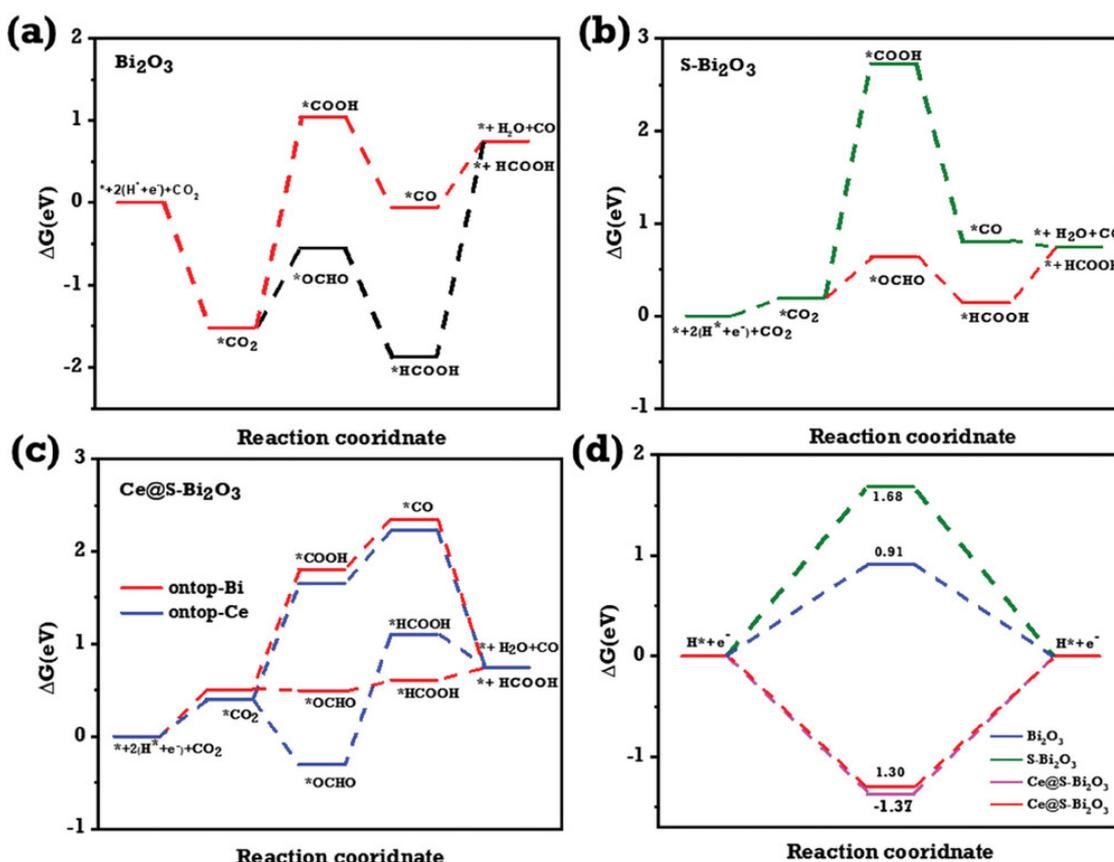
## 2.5 Theoretical Calculations

We conducted calculations to determine the optimal adsorption structures and energetics of \*CO<sub>2</sub>, \*COOH, \*OCHO, \*CO, and \*HCOOH species for the purpose of electrochemical CO<sub>2</sub> RR. These calculations were performed on the (111) plane of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub>, as shown in Figures S19 and S22 (Supporting Information). The Gibbs free energy ( $\Delta G$ ) associated with the creation of \*COOH, which serves as an intermediary in the CO pathway, is greater than that of \*OCHO, which serves as an intermediate in the HCOOH pathway in pure Bi<sub>2</sub>O<sub>3</sub> as depicted in Figure 5a. Additionally, the introduction of S-doping on the surface of Bi<sub>2</sub>O<sub>3</sub> effectively reduced the reaction route for the \*OCHO intermediate, as illustrated in Figure 5b. On S-doped Bi sites, the Gibbs free energies ( $\Delta G$ ) for creating \*OCHO and \*HCOOH through the HCOOH pathway were 0.63 and 0.14 eV, respectively.



**Figure 4**

The Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst performance and testing in three-chamber GDE reactor. a) Schematic illustration of GDE reactor cell used for the experiment. b–d) FE of products (formate, CO, and H<sub>2</sub>) distributions at different potentials applied on Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheet. e) The formate partial current density (*j*<sub>formate</sub>) for Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst at corresponding potentials in the CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolyte media.



**Figure 5**

- a) Gibbs free energy diagrams for electrochemical CO<sub>2</sub>RR to \*CO and \*HCOOH on Bi<sub>2</sub>O<sub>3</sub> (111), and b) S-Bi<sub>2</sub>O<sub>3</sub> (111) surface. c) Gibbs free energy diagrams for ECO<sub>2</sub>RR to various intermediates form on the surface of Bi (111) and Ce sites, and d) Gibbs free energy diagrams for H<sub>2</sub> on the surface of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> on top of Bi and Ce.

The introduction of S on Bi significantly reduced the Gibbs free energy for both. Furthermore, Ce and S were doped onto Bi<sub>2</sub>O<sub>3</sub>, suggesting that the \*OCHO pathway is more energetically favorable than the CO pathway, as shown in Figure 5c and Tables S3 and S4 (Supporting Information). The electrocatalyst of Ce@S-Bi<sub>2</sub>O<sub>3</sub> was established to facilitate the conversion of CO<sub>2</sub> to \*OCHO efficiently and readily with no barrier in the case of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. We also performed several models of Ce and S-doped Bi<sub>2</sub>O<sub>3</sub>, as shown in Figure S23 and Table S5 (Supporting Information). This consistently demonstrates that the reactions involving formate (\*OCHO) are endothermic, suggesting that the rate-determining step (RDS) for formate is favorable when Bi<sub>2</sub>O<sub>3</sub> is doped with Ce and S.

Furthermore, the Gibbs free energy diagram illustrating the process of hydrogen evolution reaction (HER) on the surfaces of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> is depicted in Figure 5d. The lower formation of H\* species means a higher activity of H<sub>2</sub>O dissociation on the electrocatalytic surface.<sup>[31,79]</sup> Energy barriers associated with HER on Ce@S-Bi<sub>2</sub>O<sub>3</sub> are more significant compared to those on Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub> surfaces. This indicates a significant inhibition of HER following the introduction of Ce and S-doping in Bi<sub>2</sub>O<sub>3</sub>. These calculated parameters are well agreement with experimental ECO<sub>2</sub>RR results of S and Ce modified Bi<sub>2</sub>O<sub>3</sub>

electrocatalyst (Figure 3d). In addition, the disparity in limiting potentials between the ECO<sub>2</sub>RR and HER, denoted as UL(CO<sub>2</sub>RR)-UL(H<sub>2</sub>), was computed. This parameter has been established as a crucial indicator of ECO<sub>2</sub>RR selectivity, with a more significant positive UL(CO<sub>2</sub>)-UL(H<sub>2</sub>) value corresponding to more considerable selectivity as shown in Figure S24 (Supporting Information). As anticipated, the Ce@S-Bi<sub>2</sub>O<sub>3</sub> results in a higher positive UL (CO<sub>2</sub>RR)-UL(H<sub>2</sub>) compared to both S-doped Bi<sub>2</sub>O<sub>3</sub> and pure Bi<sub>2</sub>O<sub>3</sub> electrocatalysts,<sup>[38,80]</sup> Experimental and computational results indicate that this is an impressive ECO<sub>2</sub>RR for the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst. Considering all the parameters, the Ce@S-Bi<sub>2</sub>O<sub>3</sub> group has a significant role in determining the free energy of reaction intermediates and CO<sub>2</sub> reduction activity.

### 3 Conclusion

In summary, we developed efficient Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets for ECO<sub>2</sub>RR for formate production. The obtained structural and morphological results confirmed the formation of crystalline hierarchical nanosheets. Notably, the BET results confirmed the formation of large-surface area Bi<sub>2</sub>O<sub>3</sub> (50.48 m<sup>2</sup> g<sup>-1</sup>) during S modification. The Ce and S-doped Bi<sub>2</sub>O<sub>3</sub> electrocatalyst showed improved electrocatalytic activity and long-term durability with FE<sub>formate</sub> of 90% in a wide potential window of both fabricated H-cell and GDE reactor. The obtained XPS results confirmed that the S and Ce doping induced a modification in the Bi site electronic configuration that facilitated CO<sub>2</sub> adsorption and improved electron transfer to produce a high conversion rate of formate. Moreover, DFT calculations confirmed that Ce and S optimized the adsorption of \*CO and \*OCHO intermediates in the ECO<sub>2</sub>RR. Including Ce in the S-Bi<sub>2</sub>O<sub>3</sub>, the electrocatalyst minimizes the energy barrier of intermediates on the Bi sites. Therefore, this study provides a new strategy in the development of highly efficient electrocatalysts for formate production by ECO<sub>2</sub>RR.

### 4 Experimental Section

#### *Materials*

Bismuth nitrate pentahydrate (Bi (NO<sub>3</sub>)<sub>2</sub>·5H<sub>2</sub>O, 99.99%), cerium nitrate hexahydrate (Ce (NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, 99.99%), thioacetamide (TAA,98%), ethylene glycol (EG, 99.98%), potassium bicarbonate (KHCO<sub>3</sub>, 99.7%), and Nafion solution (5 wt.%) were purchased from Sigma–Aldrich. Hydrochloric acid (HCl, 37%), sulfuric acid (H<sub>2</sub>SO<sub>4</sub>, 98%), and hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>, 35%) were obtained from Daejung Chemicals and Metals. Ethanol (EtOH, 99.5%) and acetone (99.5%) were purchased from Samchun Pure Chemical Co. Ltd. and Duksan Reagents (South Korea), respectively.

#### *Synthesis of Bi<sub>2</sub>O<sub>3</sub> and S-Bi<sub>2</sub>O<sub>3</sub>*

First, bismuth nitrate pentahydrate Bi (NO<sub>3</sub>)<sub>2</sub>·5H<sub>2</sub>O (50 mM) and thioacetamide (3 mM) were dissolved in EtOH and EG (2:1% volume) aqueous solution. The mixed solution was stirred for 30 min, transferred to a 50 mL polytetrafluoroethylene (PTFE)-lined stainless-steel autoclave, and heated at 170°C for 6 h. After hydrothermal treatment, the precipitate was collected, centrifuged, and washed with deionized water and EtOH. The obtained product was dried in a vacuum oven at 70°C overnight. The

final product was labeled as S-Bi<sub>2</sub>O<sub>3</sub>. The Bi<sub>2</sub>O<sub>3</sub> was prepared similarly except for adding thioacetamide (C<sub>2</sub>H<sub>5</sub>NS).

### ***Synthesis of Ce@S-Bi<sub>2</sub>O<sub>3</sub>***

The preparation of Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets first involved adding Bi (NO<sub>3</sub>)<sub>2</sub>·5H<sub>2</sub>O (45 mM) and Ce (NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (5 mM) into a solution containing EtOH and EG (2:1%volume). After vigorous stirring for 30 min, the solution was transferred to a 70 mL stainless steel PTFE autoclave and heated at 170 °C (heating rate 5 °C min<sup>-1</sup>) for 6 h in an air atmosphere. The products were collected and centrifuged with deionized water and EtOH several times. The obtained powder was dried in a vacuum overnight. The Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets were successfully prepared. During the synthesis, Ce and S ratios were optimized to obtain the highest electrocatalytic activity in terms of ECO<sub>2</sub>RR to formate conversions. The ratios of the highest FE were reported in the synthesis protocols.

### ***Electrochemical Analysis from an H-Cell***

An ECO<sub>2</sub>RR was performed using an electrochemical H-cell separated by a proton exchange membrane (Nafion-212). The reference electrode (Ag/AgCl) and working electrode were placed in the cathode compartment, and the counter electrode (Pt-mesh) was placed in the anode compartment. A 0.5 M KHCO<sub>3</sub> solution (pH 7.4) in electrolyte was used as a catholyte throughout the reaction. Before performing the ECO<sub>2</sub>RR, the cathode compartment was saturated with CO<sub>2</sub> (gas) (99.99% purity, BOC) by purging at 0.04 MPa (40 mL min<sup>-1</sup>) for 30 min throughout the electrolysis. All potentials measured against the reference electrode were converted to the RHE scale using  $E \text{ (vs RHE)} = E \text{ (vs. Ag/AgCl)} + 0.197 \text{ V} + 0.0591 \text{ x pH}$ . The ECO<sub>2</sub>RR activity was investigated using CA and LSV at a scan rate of 5 mV s<sup>-1</sup> at standard temperature and pressure.

### ***Electrochemical CO<sub>2</sub>RR Test in a GDE Reactor***

Quantification of the products for various electrocatalysts was performed in a three-chamber GDE reactor in a three-electrode mode described in the previous study.<sup>[78]</sup> The current set-up consists of GDE as the working electrode, Pt nanoparticle-coated Ti mesh as a counter electrode, and Ag/AgCl (3 M NaCl) as the reference electrode. In addition, 0.5 M KHCO<sub>3</sub> was utilized as an electrolyte in both electrode chambers. The electrolyte chambers were separated by a cation exchange membrane (Sustainion X37-50, Dioxide Materials) to stop cross-over of the liquid products. For the usual measurements, the reactor was operated in a batch mode. Pure CO<sub>2</sub> gas (99.99% purity, BOC) flow was kept at 20 mL min<sup>-1</sup> and was controlled by a digital mass flow controller. The current densities were determined by dividing the current obtained by the GDE area by 2 cm<sup>2</sup>.

### ***Calculation of Faradic Efficiency in a GDE Reactor***

The ECO<sub>2</sub>RR products were analyzed by GC techniques. Gas products were measured by online GC using a Shin Carbon Micropacked column (Shimadzu Nexis GC-2030), where the outlet of the gas chamber was directly connected to the injection port of the GC, and samples

were collected at regular intervals. After ECO<sub>2</sub>RR liquid products were analyzed, they were manually collected. Alcohols were analyzed using the above GC using a DB-wax column, and formate was analyzed using ion chromatography (Eco IC, Metrohm).<sup>[78]</sup>

FE refers to the ratio of current used to produce a particular product, expressed by Equation (1):

$$FE = \frac{znF}{Q} \quad (1)$$

Where  $z$  is the number of electrons required per mole of product, and  $n$  is the number of moles of product formed.  $F$  is Faraday's constant ( $96485 \text{ C mol}^{-1}$ ), and  $Q$  is the total charge passed through to the experiments. The total FE of the samples ranged from 80% to 91.5%. Gaseous products can affect the overall FE value, including side reactions of the catalytic reduction during the CO<sub>2</sub> reduction process, crossing of products through the GDE and the membrane, and averaging of the obtained currents. Total FE was normalized by 100% to allow direct comparison between the catalysts. The total voltage measured was highest in formate observed at  $-1.06 \text{ V}$  at  $41 \text{ mA cm}^{-2}$ , and the highest current density of  $54 \text{ mA cm}^{-2}$  was observed at  $-1.36 \text{ V}$  versus RHE for the Ce@S- Bi<sub>2</sub>O<sub>3</sub> electrocatalyst.

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## Conflict of Interest

The authors declare no conflict of interest.

1. J. Ni, Q. Cheng, S. Liu, M. Wang, Y. He, T. Qian, C. Yan, J. Lu, *Adv. Funct. Mater.* 2023, **33**, 2212483.
2. H. Liu, Y. Zhu, J. Ma, Z. Zhang, W. Hu, *Adv. Funct. Mater.* 2020, **30**, 1910534.
3. Álvarez, A. Bansode, A. Urakawa, A. V. Bavykina, T. A. Wezendonk, M. Makkee, J. Gascon, F. Kapteijn, *Chem. Rev.* 2017, **117**, 9804.
4. H. Shen, Y. Zhao, L. Zhang, Y. He, S. Yang, T. Wang, Y. Cao, Y. Guo, Q. Zhang, H. Zhang, *Adv. Energy Mater.* 2023, **13**, 2202818.
5. a) D. Wu, G. Huo, W. Y. Chen, X. Z. Fu, J. L. Luo, *Appl. Catal., B* 2020, **271**, 118957. b) M. Prasanna, N. Logeshwaran, S. Ramakrishnan, D. J. Yoo, *Small* 2023, **20**, 2306165.
6. Y. Hori, H. Wakebe, T. Tsukamoto, O. Koga, *Electrochim. Acta* 1994, **39**, 1833.
7. X. Zhang, Y. Zhang, Q. Li, X. Zhou, Q. Li, J. Yi, Y. Liu, J. Zhang, *J. Mater. Chem. A* 2020, **8**, 9776.

8. P. Deng, F. Yang, Z. Wang, S. Chen, Y. Zhou, S. Zaman, B. Y. Xia, *Angew. Chem., Int. Ed.* 2020, **59**, 10807.
9. Z. Wang, C. Wang, Y. Hu, S. Yang, J. Yang, W. Chen, H. Zhou, F. Zhou, L. Wang, J. Du, Y. Li, Y. Wu, *Nano Res.* 2021, **14**, 2790.
10. S. Liu, X. F. Lu, J. Xiao, X. Wang, X. W. Lou, *Angew. Chem., Int. Ed.* 2019, **58**, 13828.
11. J. Fan, X. Zhao, X. Mao, J. Xu, N. Han, H. Yang, B. Pan, Y. Li, L. Wang, Y. Li, *Adv. Mater.* 2021, **33**, 2100910.
12. Z. Wang, G. Yang, Z. Zhang, M. Jin, Y. Yin, *ACS Nano* 2016, **10**, 4559.
13. T. Montini, M. Melchionna, M. Monai, P. Fornasiero, *Chem. Rev.* 2016, **116**, 5987.
14. Y. Shi, Y. Zhou, D. R. Yang, W. X. Xu, C. Wang, F. B. Wang, J. J. Xu, X. H. Xia, H. Y. Chen, *J. Am. Chem. Soc.* 2017, **139**, 15479.
15. D. Tan, W. Lee, K. T. Park, Y. E. Jeon, J. Hong, Y. N. Ko, Y. E. Kim, *Appl. Surf. Sci.* 2023, **613**, 155944.
16. G. Wang, F. Wang, P. Deng, J. Li, C. Wang, Y. Hua, Y. Shen, X. Tian, *Mater. Rep.: Energy* 2023, **3**, 100181.
17. P. Lamagni, M. Miola, J. Catalano, M. S. Hvid, M. A. H. Mamakhel, M. Christensen, M. R. Madsen, H. S. Jeppesen, X. M. Hu, K. Daasbjerg, T. Skrydstrup, N. Lock, *Adv. Funct. Mater.* 2020, **30**, 1910408.
18. X. Wang, X. Sang, C. L. Dong, S. Yao, L. Shuai, J. Lu, B. Yang, Z. Li, L. Lei, M. Qiu, L. Dai, Y. Hou, *Angew. Chem., Int. Ed.* 2021, **60**, 11959.
19. P. Chen, T. Zhou, M. Chen, Y. Tong, N. Zhang, X. Peng, W. Chu, X. Wu, C. Wu, Y. Xie, *ACS Catal.* 2017, **7**, 7405.
20. K. Xu, H. Cheng, H. Lv, J. Wang, L. Liu, S. Liu, X. Wu, W. Chu, C. Wu, Y. Xie, *Adv. Mater.* 2018, **30**, 1703322.
21. J. Wang, S. Ning, M. Luo, D. Xiang, W. Chen, X. Kang, Z. Jiang, S. Chen, *Appl. Catal., B* 2021, **288**, 119979.
22. S. Ramakrishnan, J. Balamurugan, M. Vinothkannan, A. R. Kim, S. Sengodan, D. J. Yoo, *Appl. Catal., B* 2020, **279**, 119381.
23. T. TranPhu, R. Daiyan, Z. Fusco, Z. Ma, R. Amal, A. Tricoli, *Adv. Funct. Mater.* 2020, **30**, 1906478.
24. G. Wen, D. U. Lee, B. Ren, F. M. Hassan, G. Jiang, Z. P. Cano, J. Gostick, E. Croiset, Z. Bai, L. Yang, Z. Chen, *Adv. Energy Mater.* 2018, **8**, 1802427.
25. H. Wei, A. Tan, Z. Xiang, J. Zhang, J. Piao, Z. Liang, K. Wan, Z. Fu, *ChemSusChem* 2022, **15**, 202200752.
26. G. Zhang, X. Zheng, X. Cui, J. Wang, J. Liu, J. Chen, Q. Xu, *ACS Appl. Nano Mater.* 2022, **5**, 15465.
27. A. Zhang, Y. Liang, H. Li, X. Zhao, Y. Chen, B. Zhang, W. Zhu, J. Zeng, *Nano Lett.* 2019, **19**, 6547.
28. X. Li, N. Qian, L. Ji, X. Wu, J. Li, J. Huang, Y. Yan, D. Yang, H. Zhang, *Nanoscale Adv.* 2022, **4**, 2288.
29. J. Yan, Y. Guan, B. Marchetti, Y. Liu, F. Ning, J. Yi, X. D. Zhou, *Chem. Eng. J.* 2023, **467**, 143531.
30. J. H. Zhou, K. Yuan, L. Zhou, Y. Guo, M. Y. Luo, X. Y. Guo, Q. Y. Meng, Y. W. Zhang, *Angew. Chem., Int. Ed.* 2019, **58**, 14197.
31. W. Ma, S. Xie, X. G. Zhang, F. Sun, J. Kang, Z. Jiang, Q. Zhang, D. Y. Wu, Y. Wang, *Nat. Commun.* 2019, **10**, 892.

32. a) P. Li, F. Yang, J. Li, Q. Zhu, J. W. Xu, X. J. Loh, K. W. Huang, W. Hu, J. Lu, *Adv. Energy Mater.* 2023, **13**, 2301597. b) F. CalleVallejo, D. Loffreda, M. T. M. Koper, P. Sautet, *Nature Chemistry* 2015, **7**, 403.
33. A. Bagger, W. Ju, A. S. Varela, P. Strasser, J. Rossmeisl, *ChemPhysChem* 2017, **18**, 3266.
34. N. Han, Y. Wang, H. Yang, J. Deng, J. Wu, Y. Li, Y. Li, *Nat. Commun.* 2018, **9**, 1320.
35. Y. Mi, S. Zeng, L. Li, Q. Zhang, S. Wang, C. Liu, D. Sun, *Mater. Res. Bull.* 2012, **47**, 2623.
36. J. Bi, J. Che, L. Wu, M. Liu, *Mater. Res. Bull.* 2013, **48**, 2071.
37. Moorthy, M. R. Subramaniam, T. G. Manivasagam, D. Kumaresan, *Dalton Trans.* 2018, **47**, 8683.
38. S. Q. Liu, M. R. Gao, R. F. Feng, L. Gong, H. Zeng, J. L. Luo, *ACS Catalysis* 2021, **11**, 7604.
39. M. Zhang, W. Wei, S. Zhou, D. D. Ma, A. Cao, X. T. Wu, Q. L. Zhu, *Energy & Environm. Sci.* 2021, **14**, 4998.
40. D. Liu, X. Yan, P. Guo, Y. Yang, Y. He, J. Liu, J. Chen, H. Pan, R. Wu, *ACS Catalysis* 2023, **13**, 7698.
41. X. Cao, Y. Tian, J. Ma, W. Guo, W. Cai, J. Zhang, *Adv. Mater.* 2023, **36**, 2309648.
42. Q. Gong, P. Ding, M. Xu, X. Zhu, M. Wang, J. Deng, Q. Ma, N. Han, Y. Zhu, J. Lu, Z. Feng, Y. Li, W. Zhou, Y. Li, *Nat. Commun.* 2019, **10**, 2807.
43. a) L. Guo, F. Chen, X. Fan, W. Cai, J. Zhang, *Appl. Catal., B* 2010, **96**, 162; b) T. T. Le, V. C. Hoang, W. Zhang, J. M. Kim, J. Kim, G. Moon, S. H. Kim, *Chem. Eng. J. Adv.* 2022, **12**, 100371.
44. S. M. Yakout, *J. Environm. Chem. Eng.* 2020, **8**, 103644.
45. R. Yu, C. Qiu, Z. Lin, H. Liu, J. Gao, S. Li, Y. Yao, J. Yu, S. Yang, *ACS Mater. Lett.* 2022, **4**, 1749.
46. F. Liu, X. Ren, J. Zhao, H. Wu, J. Wang, X. Han, Y. Deng, W. Hu, *ACS Catalysis* 2022, **12**, 13533.
47. W. D. Chen, X. Q. Li, L. H. Duan, X. L. Xie, Y. D. Cui, *Appl. Surface Sci.* 1996, **100-101**, 592.
48. H. Li, G. Wang, F. Zhang, Y. Cai, Y. Wang, I. Djerdj, *RSC Adv.* 2012, **2**, 12413.
49. Y. X. Duan, Y. T. Zhou, Z. Yu, D. X. Liu, Z. Wen, J. M. Yan, Q. Jiang, *Angew. Chem., Int. Ed.* 2021, **60**, 8798.
50. R. S. Kumar, S. Prabhakaran, S. Ramakrishnan, S. C. Karthikeyan, A. R. Kim, D. H. Kim, D. J. Yoo, *Small* 2023, **19**, 2207096.
51. a) A. Hezam, K. Namratha, Q. A. Drmosh, D. Ponnamma, J. Wang, S. Prasad, M. Ahamed, C. Cheng, K. Byrappa, *ACS Appl. Nano Mater.* 2020, **3**, 138; b) D. Jiang, W. Wang, L. Zhang, Y. Zheng, Z. Wang, *ACS Catalysis* 2015, **5**, 4851.
52. P. H. Li, Z. Y. Song, M. Yang, S. H. Chen, X. Y. Xiao, W. Duan, L. N. Li, X. J. Huang, *Analyt. Chem.* 2020, **92**, 16089.
53. H. Shen, J. Bu, W. Wang, C. Wu, Y. Cao, B. Zhang, Q. Zhang, H. Zhang, *Chin. J. Chem.* 2020, **38**, 1353.
54. B. Cui, Y. Li, S. Li, Y. Xia, Z. Zheng, Y. Q. Liu, *Energy Fuels* 2020, **34**, 9932.
55. K. Masula, Y. Bhongiri, G. Raghav Rao, P. Vijay Kumar, S. Pola, M. Basude, *Opt. Mater.* 2022, **126**, 112201.
56. J. Henych, M. Št'astný, Z. Němečková, M. Kormunda, Z. Šanderová, Z. Žmudová, P. Ryšánek, Š. Stehlík, J. Ederer, M. Liegertová, J. Trögl, P. Janoš, *ACS Appl. Nano Mater.* 2022, **5**, 17956.
57. M. Prekajski, Z. Dohčević-Mitrović, M. Radović, B. Babić, J. Pantić, A. Kremenović, B. Matović, *J. Eur. Ceram. Soc.* 2012, **32**, 1983.

58. C. Silva, F. A. Sigoli, I. O. Mazali, *J. Phys. Chem. C* 2017, **121**, 12928.
59. J. Mei, T. Liao, G. A. Ayoko, Z. Sun, *ACS Appl. Mater. Interfaces* 2019, **11**, 28205.
60. S. Wu, M. Tian, Y. Hu, N. Zhang, W. Shen, J. Li, L. Guo, P. Da, P. Xi, C. H. Yan, *Inorganic Chemistry* 2023, **62**, 4088.
61. T. Tsujiguchi, Y. Kawabe, S. Jeong, T. Ohto, S. Kukunuri, H. Kuramochi, Y. Takahashi, T. Nishiuchi, H. Masuda, M. Wakisaka, K. Hu, G. Elumalai, J. Fujita, Y. Ito, *ACS Catalysis* 2021, **11**, 3310.
62. L. Fan, Z. Xia, M. Xu, Y. Lu, Z. Li, *Adv. Funct. Mater.* 2018, **28**, 1706289.
63. Z. Weng, X. Zhang, Y. Wu, S. Huo, J. Jiang, W. Liu, G. He, Y. Liang, H. Wang, *Angew. Chem., Int. Ed.* 2017, **56**, 13135.
64. R. Boppella, M. Austeria P, Y. Kim, E. Kim, I. Song, Y. Eom, D. P. Kumar, M. Balamurugan, E. Sim, D. H. Kim, T. K. Kim, *Adv. Funct. Mater.* 2022, **32**, 2202351.
65. S. Ramakrishnan, D. B. Velusamy, S. Sengodan, G. Nagaraju, D. H. Kim, A. R. Kim, D. J. Yoo, *Appl. Catal., B* 2022, **300**, 120752.
66. F. Yang, A. O. Elnabawy, R. Schimmenti, P. Song, J. Wang, Z. Peng, S. Yao, R. Deng, S. Song, Y. Lin, M. Mavrikakis, W. Xu, *Nat. Commun.* 2020, **11**, 1088.
67. D. Yao, C. Tang, A. Vasileff, X. Zhi, Y. Jiao, S. Z. Qiao, *Angew. Chem., Int. Ed.* 2021, **60**, 18178.
68. L. Lin, X. He, X. G. Zhang, W. Ma, B. Zhang, D. Wei, S. Xie, Q. Zhang, X. Yi, Y. Wang, *Angew. Chem., Int. Ed.* 2023, **62**, 202214959.
69. S. Liang, J. Xiao, T. Zhang, Y. Zheng, Q. Wang, B. Liu, *Angew. Chem., Int. Ed.* 2023, **62**, 202310740.
70. W. Ma, J. Bu, Z. Liu, C. Yan, Y. Yao, N. Chang, H. Zhang, T. Wang, J. Zhang, *Adv. Funct. Mater.* 2021, **31**, 2006704.
71. X. Ren, F. Liu, H. Wu, Q. Lu, J. Zhao, Y. Liu, J. Zhang, J. Mao, J. Wang, X. Han, Y. Deng, W. Hu, *Angew. Chem., Int. Ed.* 2024, **63**, 202316640.
72. a) X. Wang, W. Wang, B. Liu, Z. Song, Z. Ren, H. Fu, *Small* 2023, **19**, 2304084; b) Z. Gao, M. Hou, Y. Shi, L. Li, Q. Sun, S. Yang, Z. Jiang, W. Yang, Z. Zhang, W. Hu, *Chem. Sci.* 2023, **14**, 6860.
73. Y. Liu, Z. X. Lou, X. Wu, B. Mei, J. Chen, J. Y. Zhao, J. Li, H. Y. Yuan, M. Zhu, S. Dai, C. Sun, P. F. Liu, Z. Jiang, H. G. Yang, *Adv. Mater.* 2022, **34**, 2202568.
74. a) S. Ansari, M. S. Ansari, H. Devnani, S. P. Satsangee, R. Jain, *Sens. Actuat. B: Chem.* 2018, **273**, 1226. b) J. Wu, H. Xu, W. Yan, *Appl. Surface Sci.* 2016, **386**, 1.
75. H. Xiang, S. Rasul, K. Scott, J. Portoles, P. Cumpson, E. H. Yu, *J. CO<sub>2</sub> Util.* 2019, **30**, 214.
76. F. Li, L. Chen, M. Xue, T. Williams, Y. Zhang, D. R. MacFarlane, J. Zhang, *Nano Energy* 2017, **31**, 270.
77. Y. Kong, H. Hu, M. Liu, Y. Hou, V. Kolivoška, S. Vesztergom, P. Broekmann, *J. Catal.* 2022, **408**, 1.
78. P. K. Sharma, S. Rasul, D. Li, E. H. Yu, *Materials Reports: Energy* 2023, **3**, 100196.
79. T. Shinagawa, G. O. Larrazábal, A. J. Martín, F. Krumeich, J. Pérez-Ramírez, *ACS Catalysis* 2018, **8**, 837.
80. H. Shang, T. Wang, J. Pei, Z. Jiang, D. Zhou, Y. Wang, H. Li, J. Dong, Z. Zhuang, W. Chen, *Angew. Chem., Int. Ed.* 2020, **59**, 22465.

# Surface area-enhanced cerium and sulfur-modified hierarchical bismuth oxide nanosheets for electrochemical carbon dioxide reduction to formate

Naveenkumar Palanimuthu <sup>a</sup>, Mohan Raj Subramaniam <sup>a</sup>, Muthu Austeria P <sup>b</sup>, Preetam Kumar Sharma <sup>c,d</sup>, Vinoth Ramalingam <sup>e</sup>, Karthik Peramaiah <sup>f</sup>, Shanmugam Ramakrishnan <sup>a,g</sup>, Geun Ho Gu <sup>b</sup>, Eileen Hao Yu <sup>d</sup>, and Dong Jin Yoo <sup>a, h\*</sup>

<sup>a</sup> Graduate School, Department of Energy Storage/Conversion Engineering (BK21 FOUR), Hydrogen and Fuel Cell Research Center, Jeonbuk National University, Jeonju, Jeollabuk-do 54896, Republic of Korea

<sup>b</sup> Department of Energy Engineering, Korea Institute of Energy Technology (KENTECH), Naju, 58330, Republic of Korea

<sup>c</sup> Institute for Materials Discovery, University College London, Malet Place, London, WC1E 7JE, United Kingdom

<sup>d</sup> Department of Chemical Engineering, Loughborough University, Loughborough, LE11 3TU, United Kingdom

<sup>e</sup> School of Engineering, Robert Gordon University, Garthdee Road, Aberdeen AB10 7GJ, United Kingdom

<sup>f</sup> Agency for Science, Technology, and Research, Institute of Sustainability for Chemicals, Energy and Environment, 1 Pesek Road, Jurong Island, Singapore 627833, Singapore

<sup>g</sup> School of Engineering, Newcastle University, Merz Court, Newcastle upon Tyne, NE17RU, United Kingdom

<sup>h</sup> Department of Life Science, Jeonbuk National University, Jeollabuk-do 54896, Republic of Korea

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\*Corresponding Author Email ID: [djyoo@jbnu.ac.kr](mailto:djyoo@jbnu.ac.kr) (D.J. Yoo)

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## 1 Material characterization

The morphological analysis of all prepared electrocatalysts was investigated by field emission scanning electron microscope with energy-dispersive X-ray spectroscopy (EDS) with SUPRA 40 VP; Carl Zeiss, Germany) and high-resolution transmission electron microscope (HR-TEM; JEM-ARM200F, JEOL). The X-ray diffraction (XRD) pattern of the prepared electrocatalyst was measured by using PANalytical (X'PERT-PRO Powder), (model) with Cu K $\alpha$  radiation ( $\lambda = 0.154$  nm). Bi and Ce loading was evaluated by inductively coupled plasma–optical emission spectrometry (ICP-OES) with Thermo Fisher Scientific iCAP 7000 series. The chemical state of the as-obtained materials was examined by an X-ray photoelectron spectrometer (XPS; Axis-Nova, Kratos Inc.) at the Jeonju Center of the Korea Basic Science Institute (KBSI). The determination of the specific surface area and porous properties of the MG and MMGs, nitrogen adsorption-desorption isotherms were measured volumetrically at 77 K (Microtrac, BELsorp-mini II). Product quantification was analyzed via nuclear magnetic resonance spectrometers ( $^1\text{H-NMR}$ ) (AVANCE III 400, Bruker, USA) at the Future Energy Convergence Core Center (FECC).

## 2 Experimental section

### 2.1 Electrochemical active surface area

Furthermore, the electrochemical active surface area (ECSA) was calculated via the double-layer capacitance ( $C_{dl}$ ) in the non-Faradaic region and TOF obtained from the formate current density. the electrochemical surface area (ECSA) was determined by the double-layer capacitance method using the equation:

$$ECSA = C_{dl}/C_s \quad (1)$$

where  $C_{dl}$  is the double-layer capacitance measured by the cyclic voltammetry (CV) method with different scan rates at 5 to 25  $\text{mV S}^{-1}$ . The potential is from -0.8 to 1.27 V vs. RHE, and  $C_{dl}$  was determined from the plot slope of the double-layer charging current versus the scan rate.  $C_s$  is the specific capacitance whose value is  $1.12 \mu\text{F cm}^{-2}$  used in this study.

## 2.2 Calculation of TOF in H-cell system

### TOF and active sites calculation of those $\text{Bi}_2\text{O}_3$ electrocatalysts

The formate TOF per site of the  $\text{Bi}_2\text{O}_3$  catalyst was calculated using the following equations (2):

$$\text{TOF per site} = \frac{\# \text{ total formate turnovers/cm}^2 \text{ geometric area}}{\# \text{ active sites/cm}^2 \text{ geometric area}} \quad (2)$$

The total number of formate turnovers was estimated by using the following equations (3),

$$\begin{aligned} \# \text{ HCOO}^- &= \left( j \frac{\text{mA}}{\text{cm}^2} \right) \left( \frac{\frac{1\text{C}}{\text{s}}}{1000 \text{ mA}} \right) \left( \frac{1 \text{ mol e}^-}{96485 \text{ C}} \right) \left( \frac{1 \text{ mol HCOO}^-}{2 \text{ mol e}^-} \right) \left( \frac{6.022 \times 10^{23} \text{ mol HCOO}^-}{2 \text{ mol e}^-} \right) \\ &= 1.56 \times 10^{15} \frac{\text{HCOO}^-}{\text{cm}^2} \text{ per } \frac{\text{mA}}{\text{cm}^2} \end{aligned} \quad (3)$$

Further, the Ce and Bi, S content of the  $\text{Bi}_2\text{O}_3$ , S- $\text{Bi}_2\text{O}_3$ , and Ce@S- $\text{Bi}_2\text{O}_3$  catalyst was quantified using FE-SEM analysis. The  $\text{Bi}_2\text{O}_3$  at about ~34.12 wt, Accordingly, the density of active sites based on the Bi, is:

$$\begin{aligned} \# \text{ HCOO}^- &= \left( \frac{34.12}{208.98} \right) \times \frac{1 \text{ mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \\ &= (0.1632) \times \frac{1 \text{ mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \end{aligned}$$

$$= 2.94 \times 10^{18} \text{ sites cm}^{-2}$$

For example, the TOF of the catalyst at -1.16 vs RHE was evaluated below,

$$\text{TOF} = \frac{22.05 \times 3.12 \times 10^{15} \frac{\text{HCOO}^-/\text{s}}{\text{cm}^2}}{2.94 \times 10^{18} \text{ sites cm}^{-2}} = 0.0234 \text{ s}^{-1}$$

### TOF and active sites calculation of those S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts

The Formate TOF per site of the S-Bi<sub>2</sub>O<sub>3</sub> catalyst was calculated using the following equations (2):

$$\text{TOF per site} = \frac{\# \text{ total formate turnovers/cm}^2 \text{ geometric area}}{\# \text{ active sites/cm}^2 \text{ geometric area}}$$

The total number of formate turnovers was estimated by using the following equations (3),

$$\begin{aligned} \# \text{ HCOO}^- &= \left( j \frac{\text{mA}}{\text{cm}^2} \right) \left( \frac{1\text{C}}{1000 \text{ mA}} \right) \left( \frac{1\text{mol e}^-}{96485 \text{ C}} \right) \left( \frac{1\text{mol HCOO}^-}{2 \text{ mol e}^-} \right) \left( \frac{6.022 \times 10^{23} \text{mol HCOO}^-}{2\text{mol e}} \right) \\ &= 1.56 \times 10^{15} \frac{\text{HCOO}^-}{\text{cm}^2} \text{ per } \frac{\text{mA}}{\text{cm}^2} \quad (3) \end{aligned}$$

Further, Bi the and S, O content of the S-Bi<sub>2</sub>O<sub>3</sub>, catalyst was quantified using FE-SEM analysis.

The S-Bi<sub>2</sub>O<sub>3</sub> at about ~32.87 wt. Accordingly, the density of active sites based on the Bi is:

$$\begin{aligned} \# \text{ HCOO}^- &= \left( \frac{32.87}{208.980} \right) \times \frac{1\text{mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \\ &= (0.15728) \times \frac{1\text{mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \\ &= 2.8414 \times 10^{18} \text{ sites cm}^{-2} \end{aligned}$$

For example, the TOF of the catalyst at -1.16 vs RHE was evaluated below,

$$\text{TOF} = \frac{27.02 \times 3.12 \times 10^{15} \frac{\text{HCOO}^-/\text{s}}{\text{cm}^2}}{2.8414 \times 10^{18} \text{ sites cm}^{-2}} = 0.0296 \text{ s}^{-1}$$

### TOF and active sites calculation of those Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts

The Formate TOF per site of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> catalyst was calculated using the following equations (2):

$$\text{TOF per site} = \frac{\# \text{ total formate turnovers/cm}^2 \text{ geometric area}}{\# \text{ active sites/cm}^2 \text{ geometric area}}$$

The total number of formate turnovers was estimated by using the following equations (3),

$$\begin{aligned} \# \text{ HCOO}^- &= \left( j \frac{\text{mA}}{\text{cm}^2} \right) \left( \frac{\frac{1\text{C}}{\text{s}}}{1000 \text{ mA}} \right) \left( \frac{1 \text{mol e}^-}{96485 \text{ C}} \right) \left( \frac{1 \text{mol HCOO}^-}{2 \text{mol e}^-} \right) \left( \frac{6.022 \times 10^{23} \text{mol HCOO}^-}{2 \text{mol e}^-} \right) \\ &= 1.56 \times 10^{15} \frac{\text{HCOO}^-}{\text{cm}^2} \text{ per } \frac{\text{mA}}{\text{cm}^2} \end{aligned}$$

Further, Ce and Bi the S, O content of the Ce@S-Bi<sub>2</sub>O<sub>3</sub>, catalyst was quantified using FE-SEM analysis. The Ce@S-Bi<sub>2</sub>O<sub>3</sub> at about ~27.62 wt., 2.95 wt. Accordingly, the density of active sites based on the Bi and Ce is:

$$\begin{aligned} \# \text{ HCOO}^- &= \left( \frac{27.62}{208.980} + \frac{2.95}{140.116} \right) \times \frac{1 \text{mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \\ &= (0.15321) \times \frac{1 \text{mmol}}{100 \text{ mg}} \times 3 \frac{\text{mg}}{\text{cm}^2} \times 6.022 \times 10^{20} \frac{\text{sites}}{\text{mmol}} \\ &= 2.76789 \times 10^{18} \text{ sites cm}^{-2} \end{aligned}$$

For example, the TOF of the catalyst at -1.16 vs RHE was evaluated below,

$$\text{TOF} = \frac{49.9 \times 3.12 \times 10^{15} \frac{\text{HCOO}^-/\text{s}}{\text{cm}^2}}{2.76789 \times 10^{18} \text{ sites cm}^{-2}} = 0.05624 \text{ s}^{-1}$$

### 2.3 Product analysis from H-cell system

Liquid phase products were quantified using a  $^1\text{H}$ -NMR spectrometer (Avance III 400, Bruker, USA) as per previous literature.<sup>[2] [3]</sup> Initially, the standard calibration curve obtained by measuring NMR readings with different concentration HCOOH with known concentration of internal standard solvent of DMSO (peak at 2.6 ppm). After ECO<sub>2</sub>RR, (120 minutes of chronoamperometry analysis with different applied potential) the catholyte solution was collected and analyzed  $^1\text{H}$ -NMR as follows, 400  $\mu\text{L}$  of catholyte; 120  $\mu\text{L}$  of D<sub>2</sub>O; and 10  $\mu\text{L}$  of DMSO. The concentration of produced HCOOH was quantitatively analyzed by integration of the measured peak signal and calculated area ratio of DMSO and HCOOH.

$$FE(\%) = \frac{N \times n \times F}{Q_{total}} \times 100 \quad (4)$$

### 2.4 Computational methodology

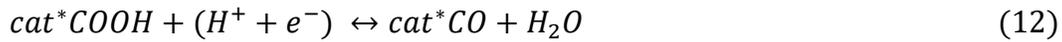
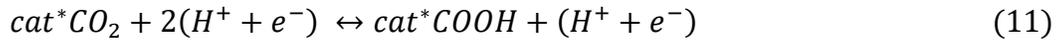
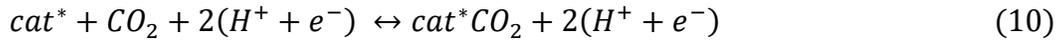
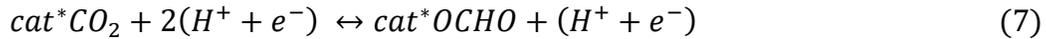
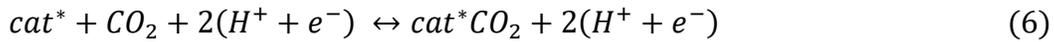
All DFT calculations were performed in Vienna ab initio simulation package,<sup>[4]</sup> using the projector augmented wave pseudopotential using a 400 eV cutoff energy for the valence electrons, and the generalized gradient approximation in the form of Perdew-Burke-Ernzerhof (PBE) for the exchange–correlation potentials.<sup>[5] [6]</sup> Van der Waals interaction was considered with the DFT-D2 method proposed by Grimme and was used to correct the dispersion 3 forces.<sup>[7]</sup> The simulation was carried out on a two-dimensional layer and vacuum regions of  $\approx 15 \text{ \AA}$  along Z-directions to avoid interaction between the layers. The Brillouin zone was sampled by  $4 \times 4 \times 1$  for optimization. The various possible active sites were considered for the adsorption of various intermediates such as CO<sub>2</sub>, COOH, OCHO, HCOOH, and CO molecules. The computational hydrogen electrode

(CHE) model<sup>8</sup> was employed to specify the Gibbs free energy of the proton-electron pair as the function of electrical potential.

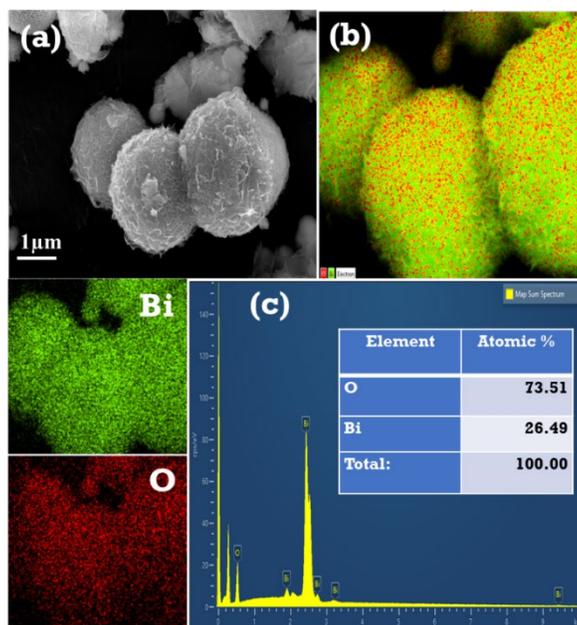
The Gibbs free energy change ( $\Delta G$ ) was calculated for the following reaction mechanism and The change in free energy is calculated by the following formula:

$$\Delta G = \Delta E + \Delta E_{ZPE} - T\Delta S \quad (5)$$

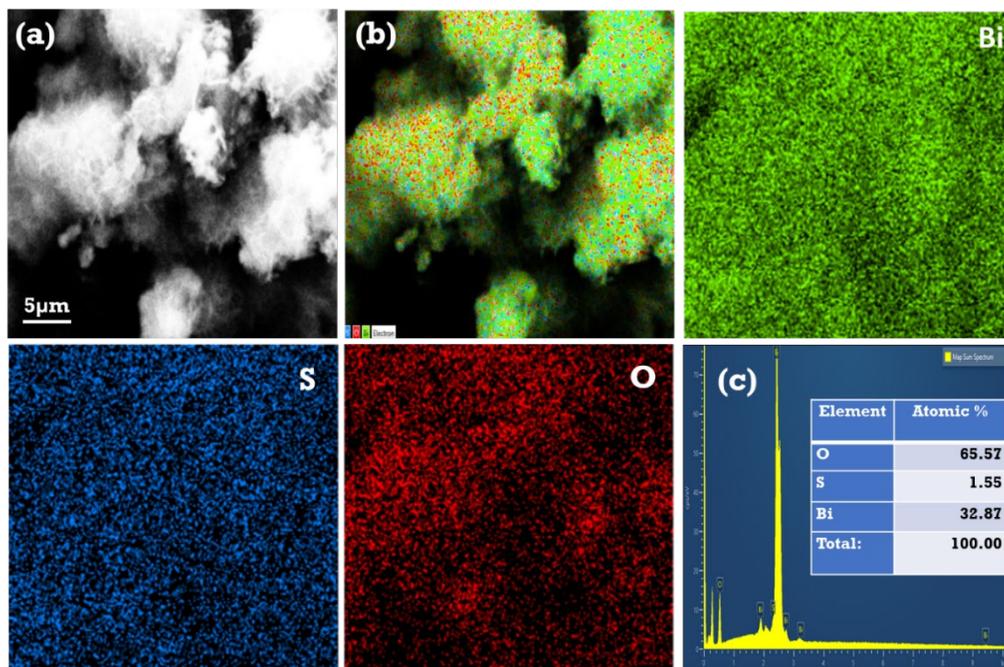
Reaction mechanisms for generating formate (6-9), CO (10-13), and H<sub>2</sub> (14-15) were assumed to be



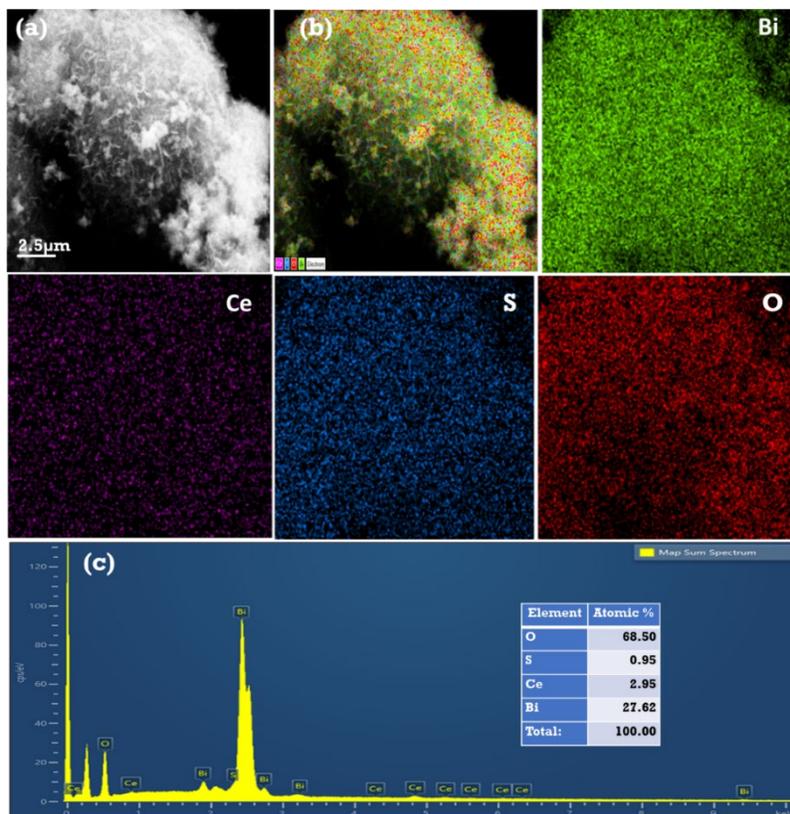
where cat\* represents either a vacant surface catalytic active site or intermediate species adsorbed on the active site.



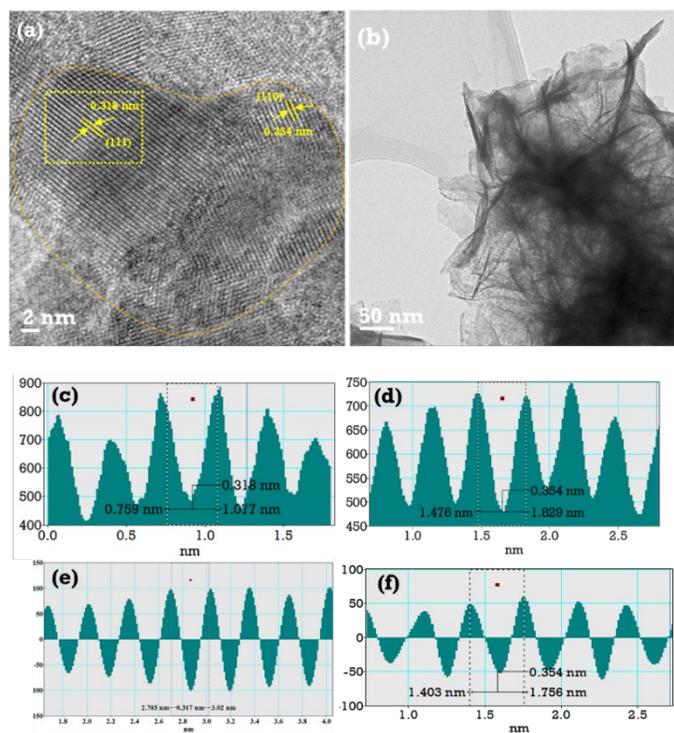
**Figure, S1(a)** illustrates the FESEM image of the Bi<sub>2</sub>O<sub>3</sub> structure. **b)** It's certainly revealing the formation of a uniform hierarchical microsphere shape of Bi<sub>2</sub>O<sub>3</sub> in the dimension of 3-4 μm. The Bi<sub>2</sub>O<sub>3</sub> microsphere structure was homogenously distributed in the overall prepared catalyst. Moreover, the magnified FESEM image confirms the formation of highly porous microspheres with hierarchical nanosheet structures. The corresponding EDS analysis confirms the presence of Bi and O elements in the overall prepared catalyst.



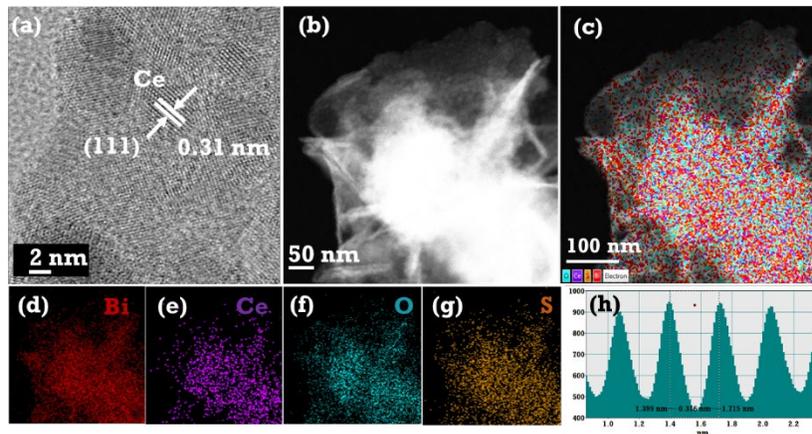
**Figure, S2 (a-c)** shows FE-SEM images and elemental mapping of sulfur-doped S-Bi<sub>2</sub>O<sub>3</sub> nanostructure. After doping with sulfur, there is not much change in the shape of the hierarchical microsphere nanosheets and the morphology of the Bi<sub>2</sub>O<sub>3</sub>. The corresponding elemental mapping and EDS analysis reveal the uniform doping of sulfur in the overall Bi<sub>2</sub>O<sub>3</sub> up to 1.2 at%.



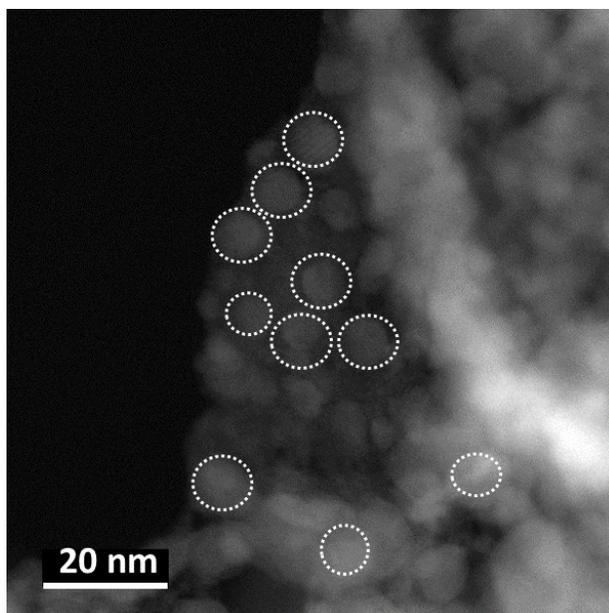
**Figure, S3 (a-b)** illustrates the FE-SEM micrographs of cerium-doped S-Bi<sub>2</sub>O<sub>3</sub> and their corresponding elemental mapping. It confirmed the doping of the cerium atom with S-Bi<sub>2</sub>O<sub>3</sub> without delaminating the hierarchical nanosheet morphology of the Bi<sub>2</sub>O<sub>3</sub> structure. The corresponding element mapping and EDS analysis show the uniform distribution of elements Bi, Ce, O, and S throughout the catalyst. **c)** represents the low intensities of the S and Ce signals indicating a relatively low content of these elements present in the catalyst.



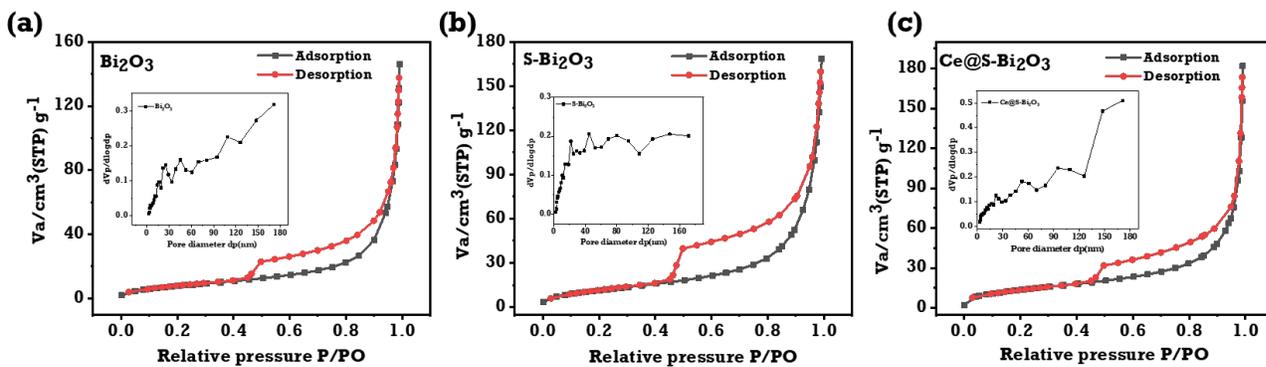
**Figure, S4 (a-b)** HR-TEM and TEM images of Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets. **(c-f)** reveals the FFT pattern and IFFT pattern of the corresponding catalyst for ECO<sub>2</sub>RR.



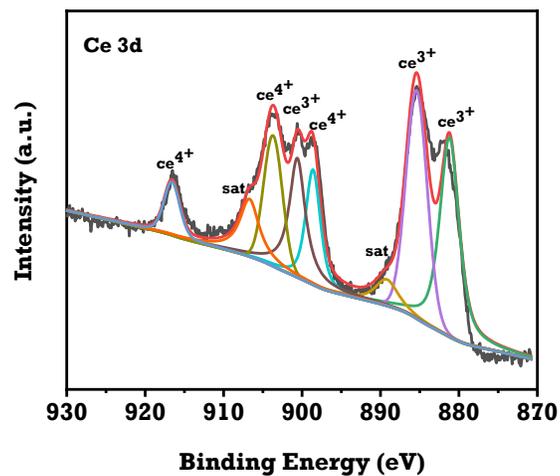
**Figure, S5 (a-b)** illustrates the HAADF-STEM, and elemental color mapping of Ce@S-Bi<sub>2</sub>O<sub>3</sub> hierarchical nanosheets. **(c-h)** The overall elemental mapping and corresponding EDS elemental mapping with FFT image.



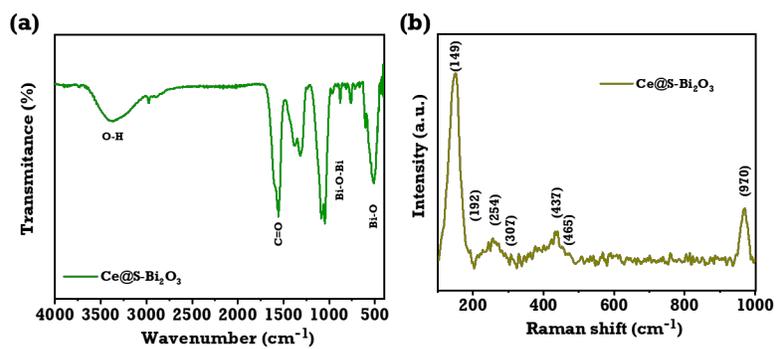
**Figure, S6** illustrates the (HAADF-STEM) of Ce@S-Bi<sub>2</sub>O<sub>3</sub>, it further identifies the structure of Bi nanosheets showing that the nanoparticles are making well-defined flexible hierarchical nanosheets appear.



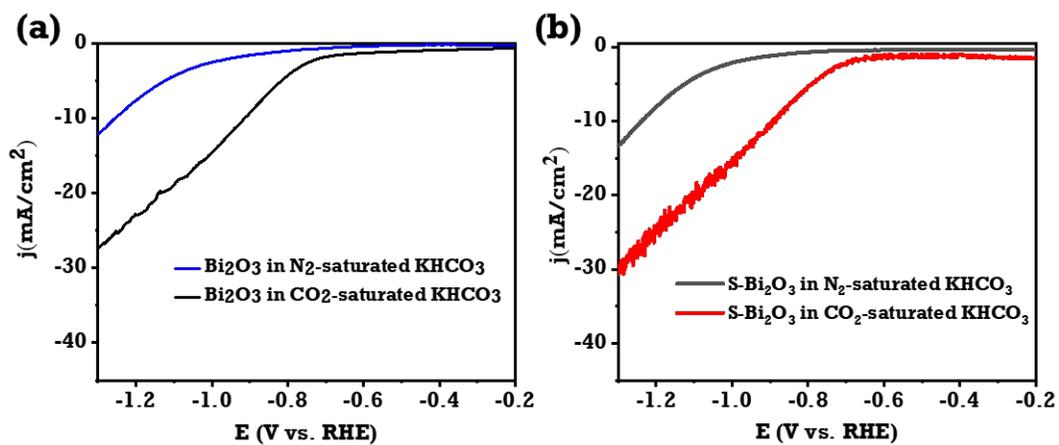
**Figure, S7 (a-c)** Nitrogen adsorption - desorption isotherms and pore size distributions of  $\text{Bi}_2\text{O}_3$ ,  $\text{S-Bi}_2\text{O}_3$ , and  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst.



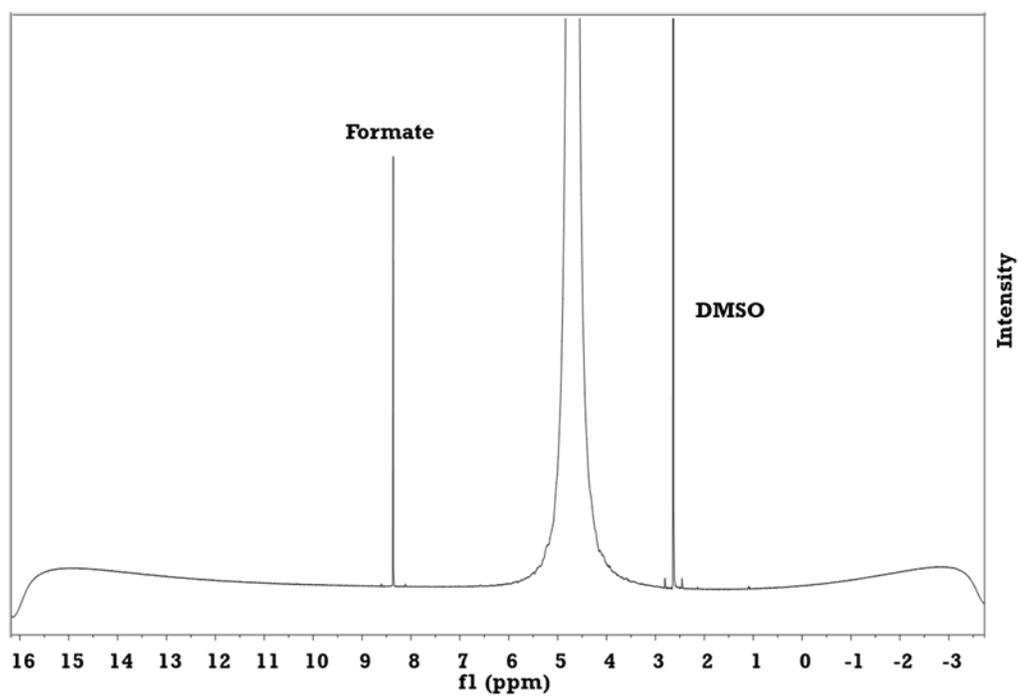
Figure, S8 represents the XPS spectrum of core Ce 3d.



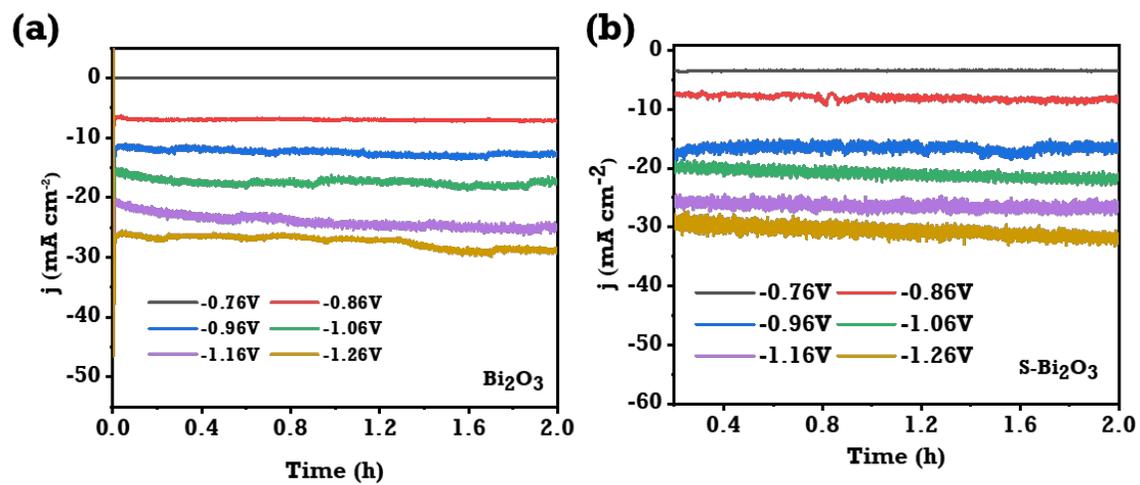
Figure, S9 (a-b) the FT-IR and Raman spectrum of Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst



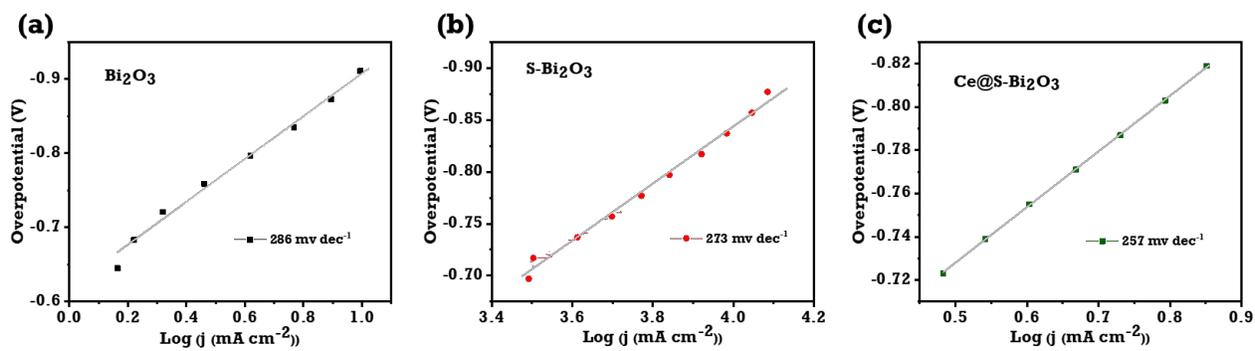
Figure, S10 (a-b) shows a LSV curve of obtained  $\text{Bi}_2\text{O}_3$ ,  $\text{S-Bi}_2\text{O}_3$ , and  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst.



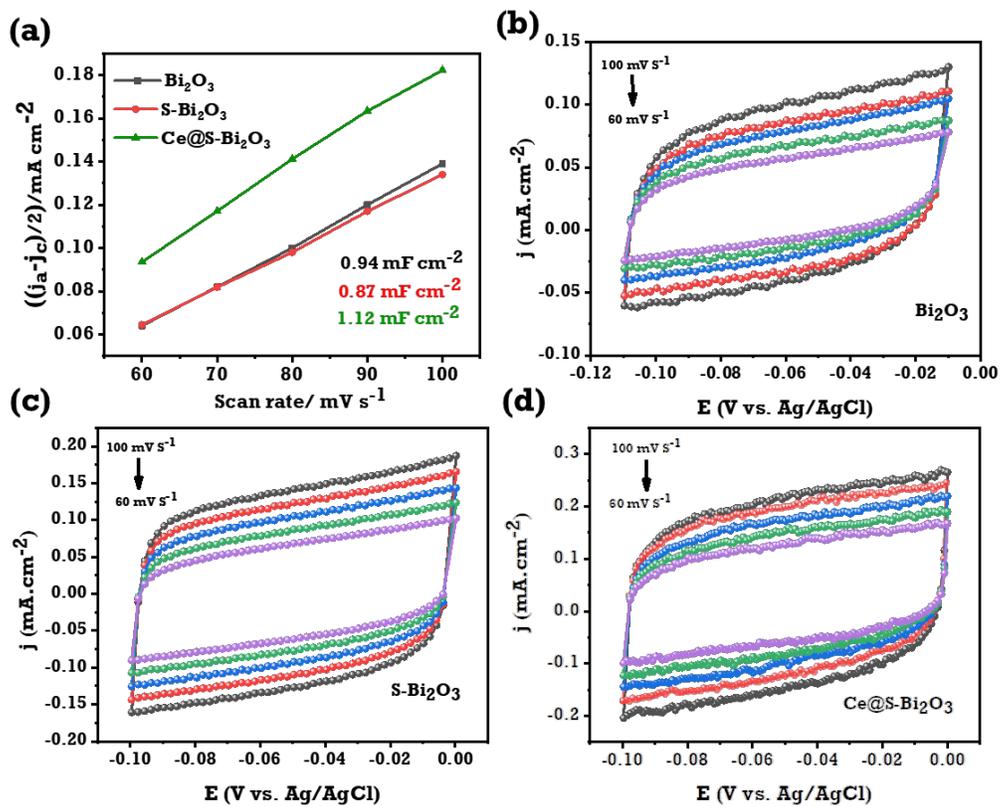
**Figure, S11**  $^1\text{H-NMR}$  analysis of formate with DMSO as an internal standard obtained from the  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalyst.



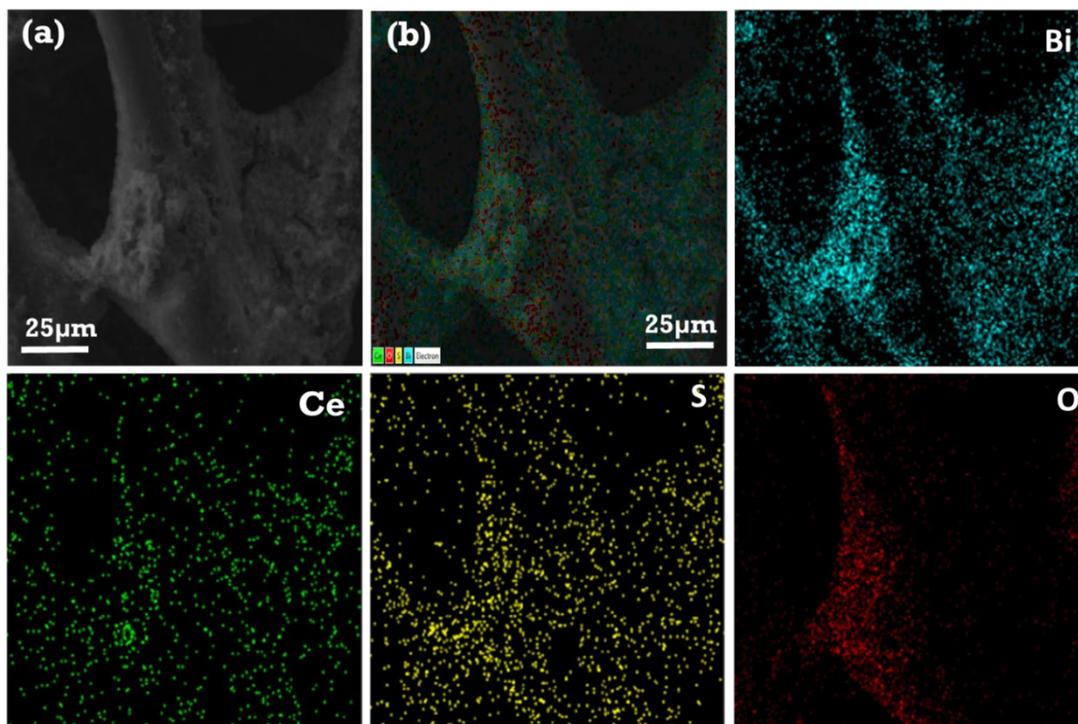
Figure, S12 (a-b) shows the chronoamperometry of corresponding electrocatalysts of  $\text{Bi}_2\text{O}_3$  and  $\text{S-Bi}_2\text{O}_3$ .



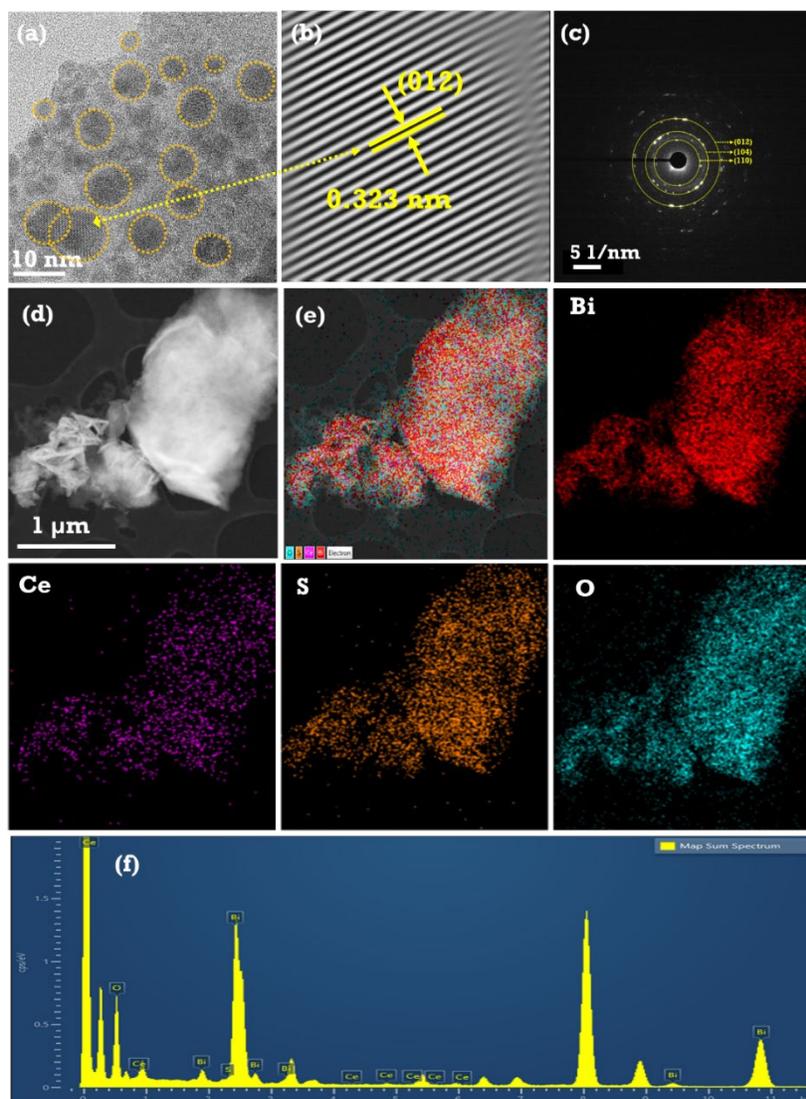
Figure, S13 (a-c) Tafel slope of Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalysts.



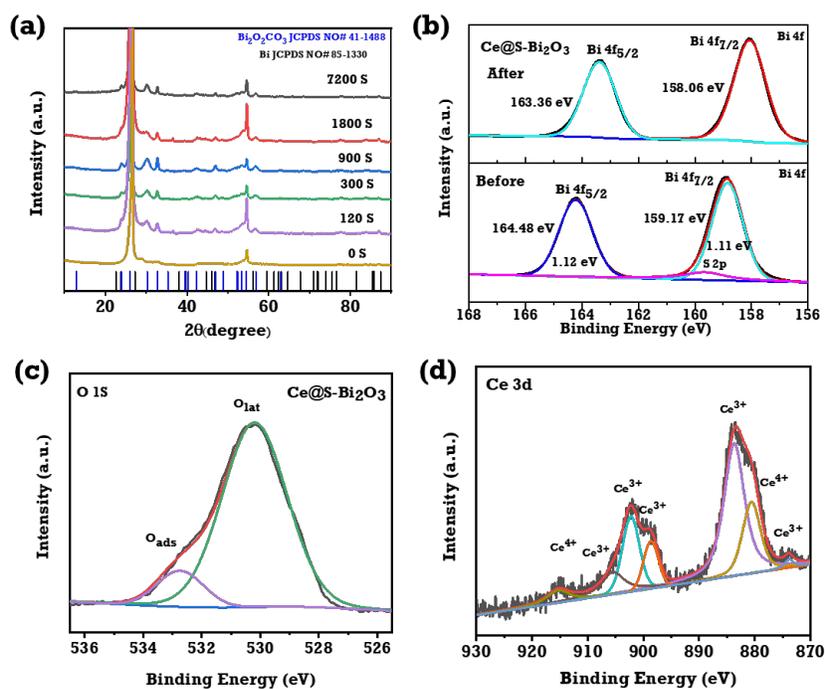
**Figure, S14 (a-d)** Double layer capacitance ( $C_{dl}$ ) measurements curve of  $\text{Bi}_2\text{O}_3$ ,  $\text{S-Bi}_2\text{O}_3$ , and  $\text{Ce@S-Bi}_2\text{O}_3$  electrocatalysts.



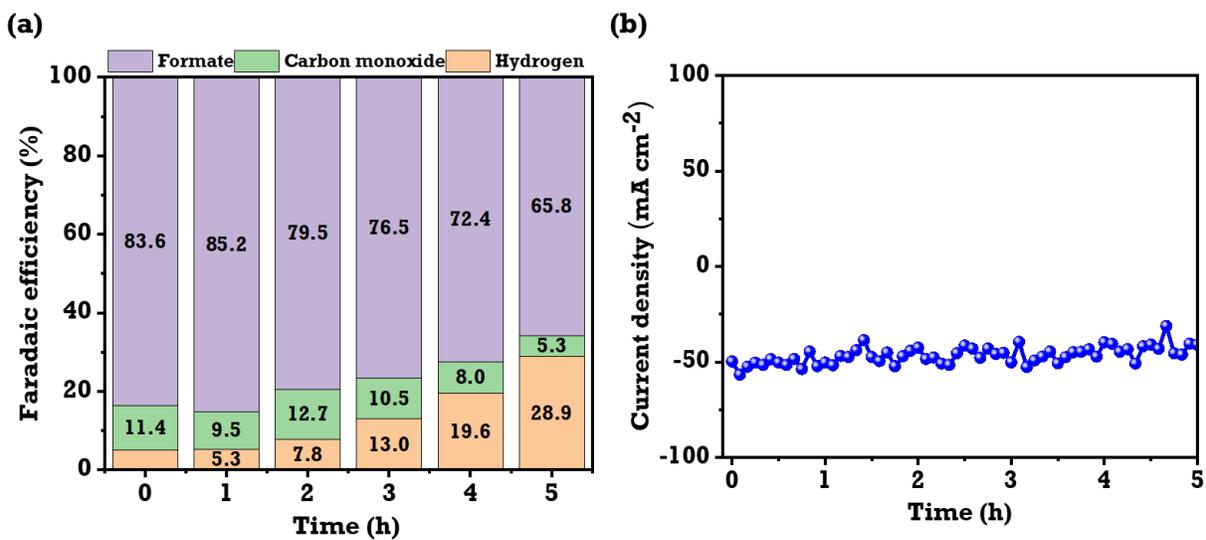
**Figure, S15(a)** After the  $\text{ECO}_2\text{RR}$  the FE-SEM images show a  $\text{Ce@S-Bi}_2\text{O}_3$  catalyst. **(b)** reveals the overlap of catalysts in the present electrode areas and the evidence of respective EDS elemental mapping of the catalysts was presents Bi, Ce, S, O the corresponding elemental mapping.



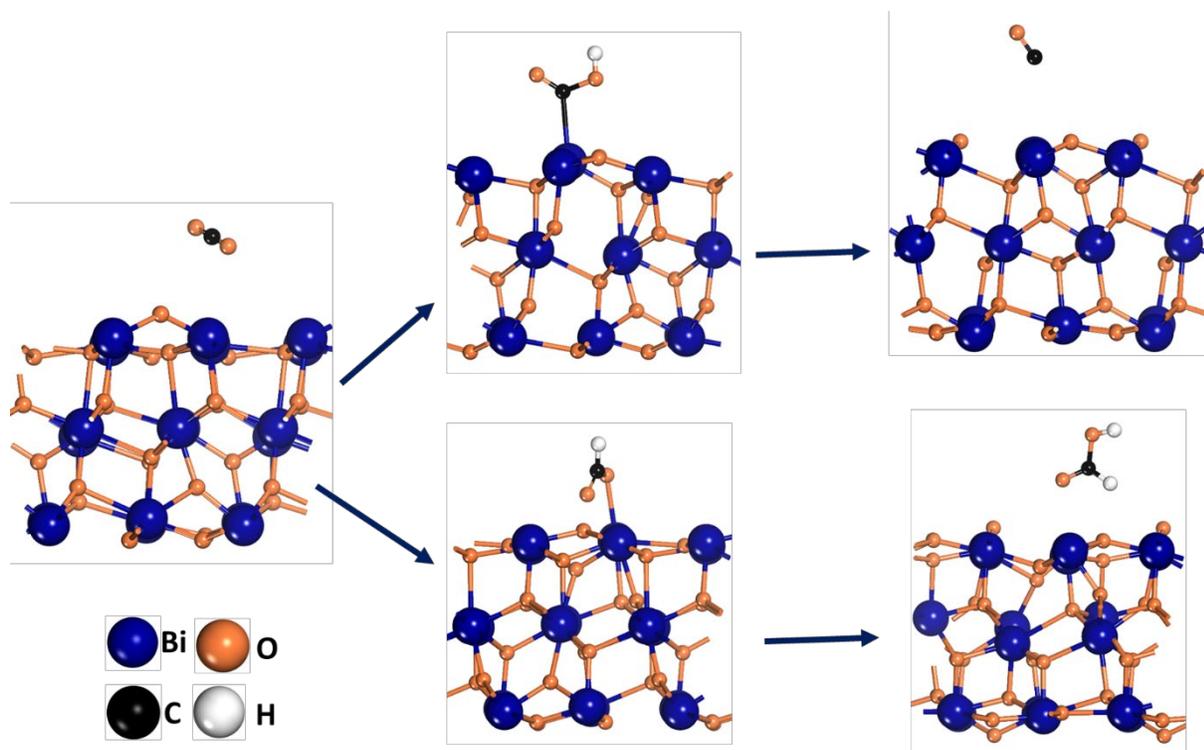
**Figure, S16 (a-c)** represents the HRTEM images of Ce@S-Bi<sub>2</sub>O<sub>3</sub> and the corresponding d-spacing plane and SAED patterns with crystal planes. **(d-f)** shows the HR-TEM image and the corresponding EDS mapping spectrum after the ECO<sub>2</sub>RR performance.



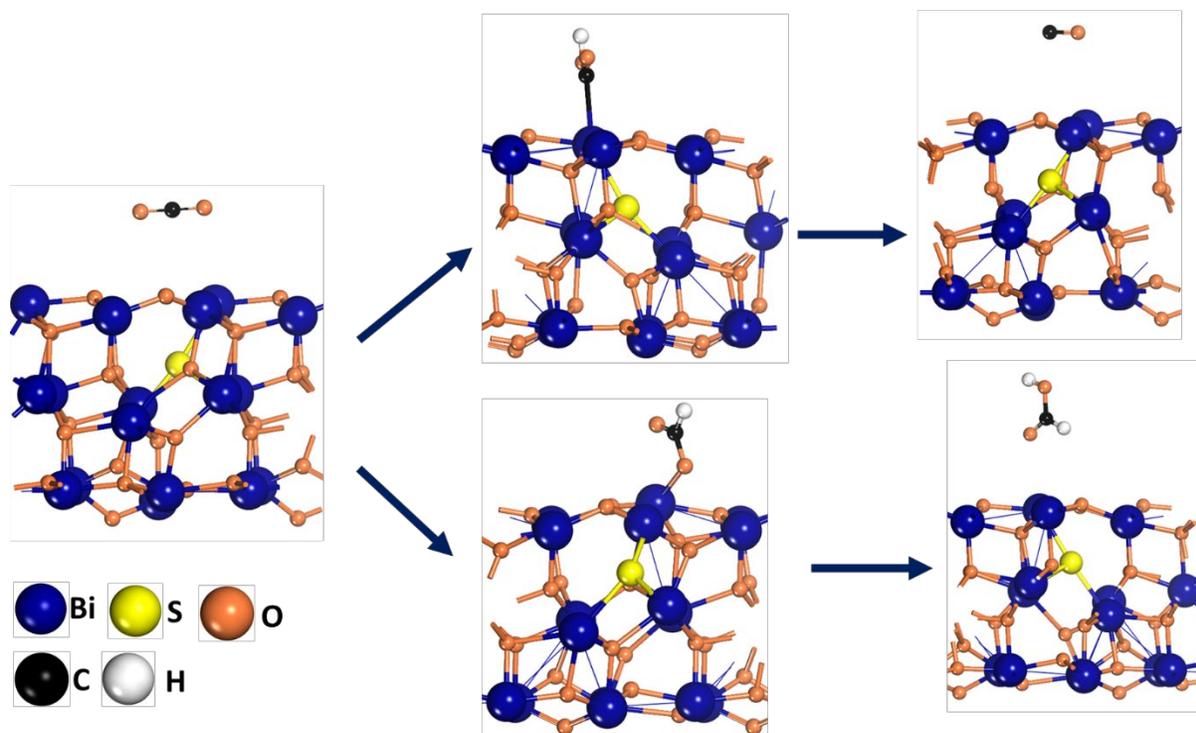
**Figure, S17 (a)** Time-dependent XRD curve of the Ce@S-Bi<sub>2</sub>O<sub>3</sub> electrocatalyst during the ECO<sub>2</sub>RR. **(b-d)** Before and after (2 h) electrochemical CO<sub>2</sub>RR shows the XPS spectra of Bi4f and O1s, Ce 3d survey spectrum of Ce@S-Bi<sub>2</sub>O electrocatalysts.



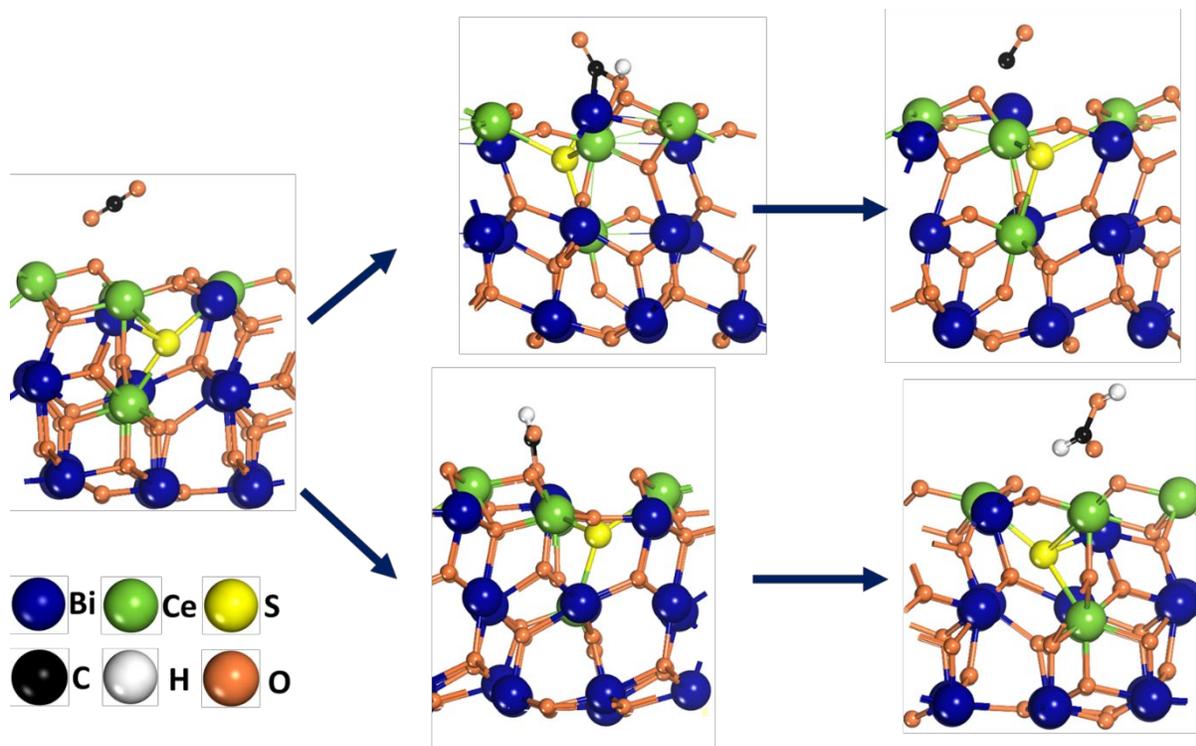
**Figure S18** (a) Faradaic efficiency, (b) current density of Ce@S-Bi<sub>2</sub>O<sub>3</sub> at -1.2 V vs RHE during the continuous stability test up to 5 hours.



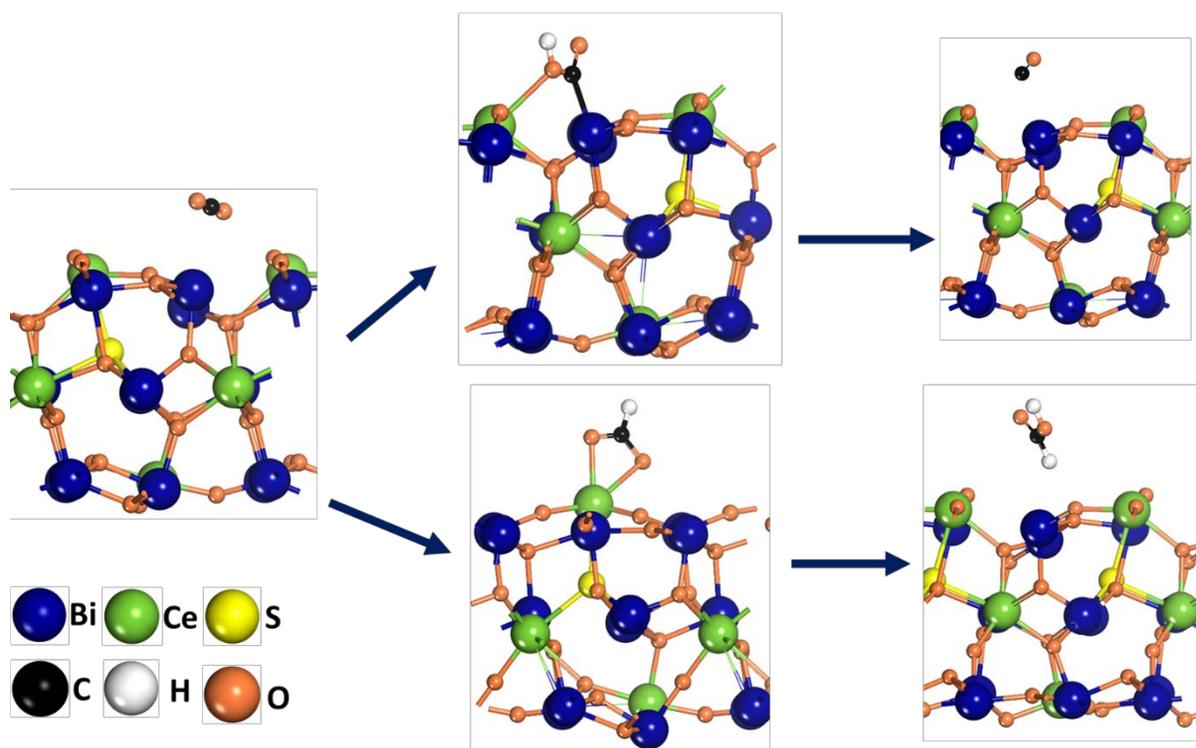
**Figure, S19** Optimized configurations of \*CO<sub>2</sub>, \*COOH, \*CO, \*OCHO, and \*HCOOH on (111) planes of Bi<sub>2</sub>O<sub>3</sub>.



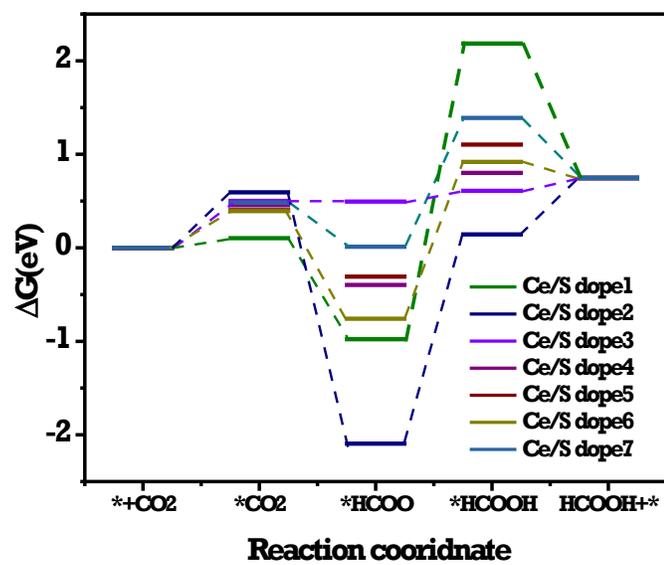
**Figure, S20** Optimized configurations of  $*\text{CO}_2$ ,  $*\text{COOH}$ ,  $*\text{CO}$ ,  $*\text{OCHO}$ , and  $*\text{HCOOH}$  on (111) planes of S doped on  $\text{Bi}_2\text{O}_3$ .



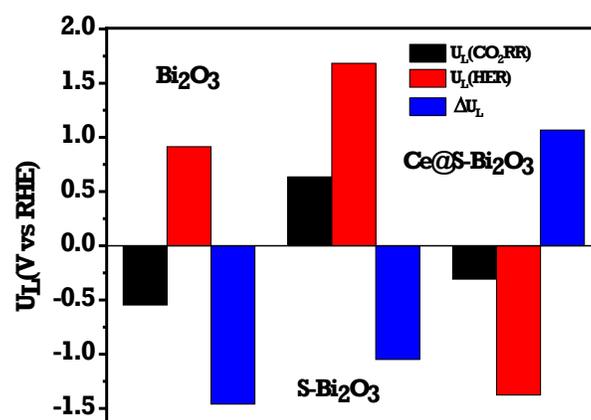
**Figure, S21** Optimized configurations of \*CO<sub>2</sub>, \*COOH, \*CO, \*OCHO, and \*HCOOH on (111) planes of Ce@S-doped on Bi<sub>2</sub>O<sub>3</sub> (On top of Bi).



**Figure, S22** Optimized configurations of \*CO<sub>2</sub>, \*COOH, \*CO, \*OCHO, and \*HCOOH on (111) planes of Ce@S-doped on Bi<sub>2</sub>O<sub>3</sub> (On top of Ce).



Figure, S23 shows several models for Ce and S doping in  $\text{Bi}_2\text{O}_3$



**Figure, S24** Limiting potential difference between CO<sub>2</sub>RR and HER for Bi<sub>2</sub>O<sub>3</sub>, S-Bi<sub>2</sub>O<sub>3</sub>, and Ce@S-Bi<sub>2</sub>O<sub>3</sub> at each site.

**Table S1:** BET surface area, pore volume, and diameter of electrocatalysts

Materials	BET surface area (m <sup>2</sup> g <sup>-1</sup> )	Pore volume (cm <sup>3</sup> g <sup>-1</sup> )	Pore diameter (nm)
Bi <sub>2</sub> O <sub>3</sub>	30.403	0.2129	28.005
S-Bi <sub>2</sub> O <sub>3</sub>	44.019	0.2447	22.232
Ce@S-Bi <sub>2</sub> O <sub>3</sub>	50.482	0.2716	21.522

**Table S2:** The ICP-OES results of Ce@S-Bi<sub>2</sub>O<sub>3</sub> (At. %) electrocatalyst before and after ECO<sub>2</sub>RR.

Ce@S-Bi <sub>2</sub> O <sub>3</sub> Electrocatalyst	Bi (At. %)	Ce (At. %)	S (At. %)
Before ECO <sub>2</sub> RR	27.8	3.3	1.2
After ECO <sub>2</sub> RR	17.0	2.5	0.8

**Table S3:** Gibbs free energy changes ( $\Delta G$ ) in the reaction steps formation of HCOOH

materials	$\Delta G1(*CO_2)$	$\Delta G2(*OCHO)$	$\Delta G3(*HCOOH)$
Bi <sub>2</sub> O <sub>3</sub>	-1.52099	-0.54595	-1.87043
S-Bi <sub>2</sub> O <sub>3</sub>	0.19033	0.63571	0.14487
Ce/S- Bi <sub>2</sub> O <sub>3</sub>	0.50134	0.49527	0.60952
Ce/S- Bi <sub>2</sub> O <sub>3</sub>	0.4032	-0.3081	1.10412

**Table S4:** Gibbs free energy changes ( $\Delta G$ ) in the reaction steps formation of CO

materials	$\Delta G1(*CO_2)$	$\Delta G2(*COOH)$	$\Delta G3(*CO)$
Bi <sub>2</sub> O <sub>3</sub>	-1.52099	1.03843	-0.05979
S-Bi <sub>2</sub> O <sub>3</sub>	0.19033	2.72785	0.79759
Ce/S- Bi <sub>2</sub> O <sub>3</sub>	0.50134	1.8057	2.3497
Ce/S- Bi <sub>2</sub> O <sub>3</sub>	0.4032	1.65222	2.22612

**Table S5:** Gibbs free energy changes ( $\Delta G$ ) in the reaction steps formation of HCOOH for Ce/S doping in Bi<sub>2</sub>O<sub>3</sub>

materials	$\Delta G1(*CO_2)$	$\Delta G2(*OCHO)$	$\Delta G3(*HCOOH)$
Ce/S- dope1	0.10336	-0.9782	2.18534
Ce/S- dope2	0.59413	-2.09397	0.14357
Ce/S- dope3	0.50134	0.49527	0.60952
Ce/S- dope4	0.45647	-0.3976	0.80099
Ce/S- dope5	0.4032	-0.3081	1.10412
Ce/S- dope6	0.39233	-0.75867	0.91896
Ce/S- dope7	0.48238	0.01174	1.38861

**Table S6:** Performance comparison of various Bi-electrocatalyst ECO<sub>2</sub> RR in H-cell.

Materials	Cell configuration	electrolyte	Potential (V vs. RHE)	J <sub>formate</sub> (mA cm <sup>-2</sup> )	FE <sub>formate</sub>	Ref
Ce@S-Bi <sub>2</sub> O <sub>3</sub>	H-cell	0.5 M KHCO <sub>3</sub>	-1.16	±42.09	± 92.5%	<b>This work</b>
Bi /Bi <sub>2</sub> O <sub>3</sub> -CP	H-cell	0.5 M KHCO <sub>3</sub>	-1.17	72	90.4	[8]
Cux-Bi/Bi <sub>2</sub> O <sub>3</sub> @C	H-cell	0.5 M KHCO <sub>3</sub>	-0.94	10.1	93	[9]
Bi-NAs	H-cell	0.5 M KHCO <sub>3</sub>	-0.95	45	90	[10]
Bi /Bi <sub>2</sub> O <sub>3</sub>	H-cell	0.5 M KHCO <sub>3</sub>	-1.3	87.14	90	[11]
Bi-PVP/CC600	H-cell	0.5 M KHCO <sub>3</sub>	-0.83	54	86	[12]
Bi-MOF	Home-made three-electrode cell	0.5 M KHCO <sub>3</sub>	-0.9	41.0	92.2	[13]
OD-BiNSs	H-cell	0.5 M KHCO <sub>3</sub>	-0.95	62	93	[14]
Bi nanostructure	H-cell	0.5 M KHCO <sub>3</sub>	-0.9	18	92	[15]
Bi <sub>2</sub> O <sub>2</sub> CO <sub>3</sub>	H-cell	0.5M NaHCO <sub>3</sub>	-0.7	11	95	[16]
Bi nanotubes	H-cell	0.1M KHCO <sub>3</sub>	-1.0	22	95	[17]
Bi <sub>2</sub> S <sub>3</sub> -Bi <sub>2</sub> O <sub>3</sub> @rGO	H-cell	0.5M KHCO <sub>3</sub>	-0.9	3.5	90	[18]
Bi-Sn/CF	H-cell	0.5M KHCO <sub>3</sub>	-1.14	43.2	96	[19]
S- Bi <sub>2</sub> O <sub>3</sub> -CNT	H-cell	0.5M KHCO <sub>3</sub>	-0.9	28.17	97.06	[20]
Bi <sub>2</sub> O <sub>3</sub> @C	H-cell	0.5M KHCO <sub>3</sub>	-0.9	7.5	92	[21]
Bi <sub>2</sub> O <sub>3</sub> NP	H-cell	0.5M KHCO <sub>3</sub>	-0.83	3.4	92	[22]

## Reference:

- [1] T. Chatterjee, E. Boutin, M. Robert, *Dalton Transactions* **2020**, 49, 4257.
- [2] C. S. Chen, A. D. Handoko, J. H. Wan, L. Ma, D. Ren, B. S. Yeo, *Catalysis Science & Technology* **2015**, 5, 161.
- [3] D. Ren, J. Gao, L. Pan, Z. Wang, J. Luo, S. M. Zakeeruddin, A. Hagfeldt, M. Grätzel, *Angewandte Chemie International Edition* **2019**, 58, 15036.
- [4] a) G. Kresse, J. Furthmüller, *Computational Materials Science* **1996**, 6, 15  
b) G. Kresse, J. Furthmüller, *Physical Review B* **1996**, 54, 11169.
- [5] J. P. Perdew, M. Ernzerhof, K. Burke, *The Journal of Chemical Physics* **1996**, 105, 9982.
- [6] J. P. Perdew, K. Burke, M. Ernzerhof, *Physical Review Letters* **1996**, 77, 3865.
- [7] S. Grimme, *Journal of Computational Chemistry* **2006**, 27, 1787.
- [8] D. Wu, G. Huo, W. Y. Chen, X. Z. Fu, J. L. Luo, *Applied Catalysis B: Environmental* **2020**, 271, 118957.
- [9] Y. Xue, C. Li, X. Zhou, Z. Kuang, W. Zhao, Q. Zhang, H. Chen, *ChemElectroChem* **2022**, 9, e202101648.
- [10] J. Fan, X. Zhao, X. Mao, J. Xu, N. Han, H. Yang, B. Pan, Y. Li, L. Wang, Y. Li, *Advanced Materials* **2021**, 33, 2100910.
- [11] W. Wang, G. Ruan, X. Wang, C. Wu, Q. Wang, *New Journal of Chemistry* **2023**, 47, 8894.
- [12] D. Wu, X. Wang, X.Z. Fu, J.L. Luo, *Applied Catalysis B: Environmental* **2021**, 284, 119723.
- [13] F. Li, G. H. Gu, C. Choi, P. Kolla, S. Hong, T.S. Wu, Y.L. Soo, J. Masa, S. Mukerjee, Y. Jung, J. Qiu, Z. Sun, *Applied Catalysis B: Environmental* **2020**, 277, 119241.
- [14] J. Lee, H. Liu, Y. Chen, W. Li, *ACS Applied Materials & Interfaces* **2022**, 14, 14210.

- [15] P. Lu, D. Gao, H. He, Q. Wang, Z. Liu, S. Dipazir, M. Yuan, W. Zu, G. Zhang, *Nanoscale* **2019**, 11, 7805.
- [16] Y. Zhang, X. Zhang, Y. Ling, F. Li, A. M. Bond, J. Zhang, *Angewandte Chemie International Edition* **2018**, 57, 13283.
- [17] K. Fan, Y. Jia, Y. Ji, P. Kuang, B. Zhu, X. Liu, J. Yu, *ACS Catalysis* **2020**, 10, 358.
- [18] X. Yang, P. Deng, D. Liu, S. Zhao, D. Li, H. Wu, Y. Ma, B. Y. Xia, M. Li, C. Xiao, S. Ding, *Journal of Materials Chemistry A* **2020**, 8, 2472.
- [19] G. Wen, D. U. Lee, B. Ren, F. M. Hassan, G. Jiang, Z. P. Cano, J. Gostick, E. Croiset, Z. Bai, L. Yang, Z. Chen, *Advanced Energy Materials* **2018**, 8, 1802427.
- [20] S.Q. Liu, M.-R. Gao, R.F. Feng, L. Gong, H. Zeng, J.L. Luo, *ACS Catalysis* **2021**, 11, 7604.
- [21] P. Deng, F. Yang, Z. Wang, S. Chen, Y. Zhou, S. Zaman, B. Y. Xia, *Angewandte Chemie International Edition* **2020**, 59, 10807.
- [22] C.C. Miao, G.Q. Yuan, *ChemElectroChem* **2018**, 5, 3741.